

Large scale electronic structure calculation theory and its application

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1. Introduction

Problem in Electronic Structure Calculations, Multi-scale

2. Mathematics in Large-scale Calculations

2-1. Wannier state representation and Order-N method

2-2. Krylov subspace method

2-3. Hybrid method and parallelization

3. Examples

3-1. Fracture Propagation and Surface Creation in Si

3-2. Carbon Nanotube

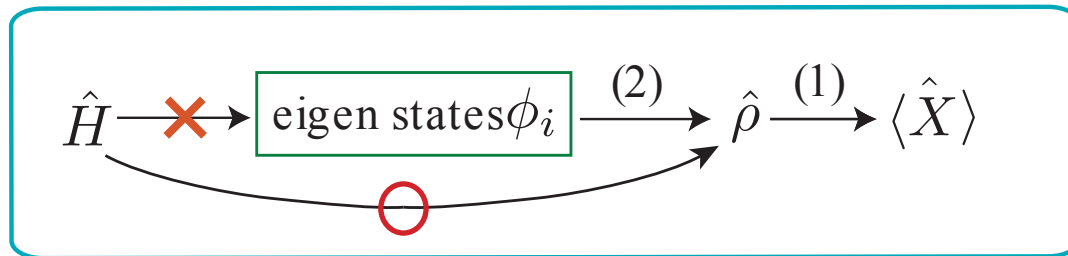
3-3. Gold Helical Multishell Nanotube

4. Conclusions

Large-scale electronic-structure calculations

Why is it difficult for large systems?

Because electronic wavefunctions are not localized and the CPU resource should be proportional to N^3



Can we get the density matrix without a knowledge on eigenstates?

Yes, we can get ρ without ϕ .

Calculating a physical quantity $\langle X \rangle$

$$\langle \hat{X} \rangle = \sum_i^{\text{occ.}} \langle \phi_i | \hat{X} | \phi_i \rangle = \text{Tr}[\hat{\rho} \hat{X}] \quad (1)$$

$$\hat{\rho} \equiv \sum_i^{\text{occ.}} |\phi_i\rangle \langle \phi_i| \quad (2)$$

:(one-body) density matrix

Wannier representation.

Hoshi and Fujiwara

J. Phys. Soc. Jpn **69**, 3773 (2000)

Order-N

Hoshi and Fujiwara

J. Phys. Soc. Jpn **72**, 2880 (2003)

Krylov (Subspace Diagonalization)

Takayama, Hoshi and Fujiwara

J. Phys. Soc. Jpn **73**, 1519 (2004)

Nano-scale cleavage in Si

Hoshi, Iguchi, Fujiwara

Phys. Rev. B **72**, 075323 (2005)

Krylov (Shifted COCG)

Takayama, Hoshi and Fujiwara

Phys. Rev. B **73**, 165108 (2006)

Hybrid Method

Hoshi and Fujiwara

J. Phys. Cond. Mat. **18**, 10787 (2006)

Our Large-scale Electronic Structure Calculations

Usually
hybrid method with
quantum mechanical MD (in small region)
and
classical mechanical MD (in larger region)

Our method
hybrid method with
different kinds of quantum mechanical MDs (over whole region)

- *nonlinearity due to shape, force, inhomogeneous contact, wavefunctions*
- *non-equilibrium, transient process*

Wannier state method without using eigen states

$$H_{\text{eff}}\psi_i = \sum_{j=1}^{\text{occ.}} \varepsilon_{ij}\psi_j$$

$$\psi_j = \sum_k^{\text{occ.}} U_{jk}\psi_k^{(\text{eig})}$$

$$\langle \hat{X} \rangle \equiv \sum_k^{\text{occ}} \langle \psi_k^{(\text{eig})} | \hat{X} | \psi_k^{(\text{eig})} \rangle = \sum_j^{\text{occ}} \langle \psi_j | \hat{X} | \psi_j \rangle$$

→ "generalized Wannier states",
 "ab initio bonding orbitals"

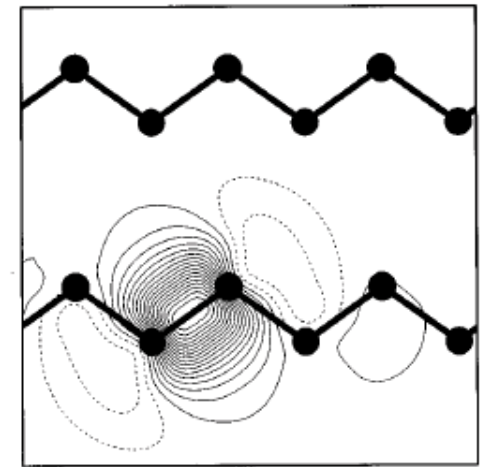
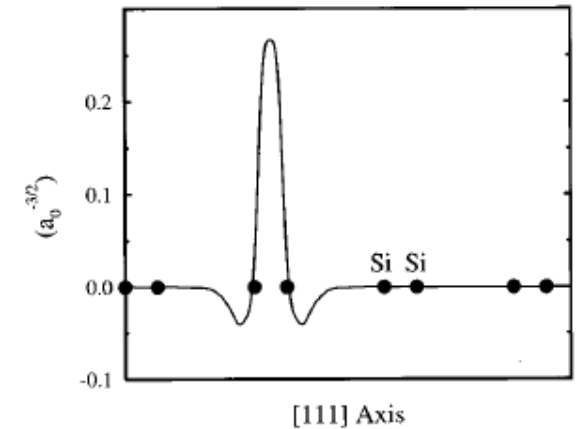
Many ab initio applications → Exact, but not order-N

$$(1) \quad |\phi_j\rangle \approx C^{(0)}|b_j\rangle + \sum_{i(\neq j)} C^{(\nu(i))}|a_i\rangle. \quad \frac{C^{(\nu)}}{C^{(0)}} = \frac{\langle a^{(\nu)} | H | b_k \rangle}{\varepsilon_b - \varepsilon_a}.$$

$$(2) \quad H_{\text{WS}}^{(k)}\psi_k = \varepsilon_{kk}\psi_k, \quad H_{\text{WS}}^{(k)} = H - \bar{\rho}_k\Omega - \Omega\bar{\rho}_k$$

$$\bar{\rho}_k = \rho - |\psi_k\rangle\langle\psi_k| = \sum_{j \neq k}^N |\psi_j\rangle\langle\psi_j|$$

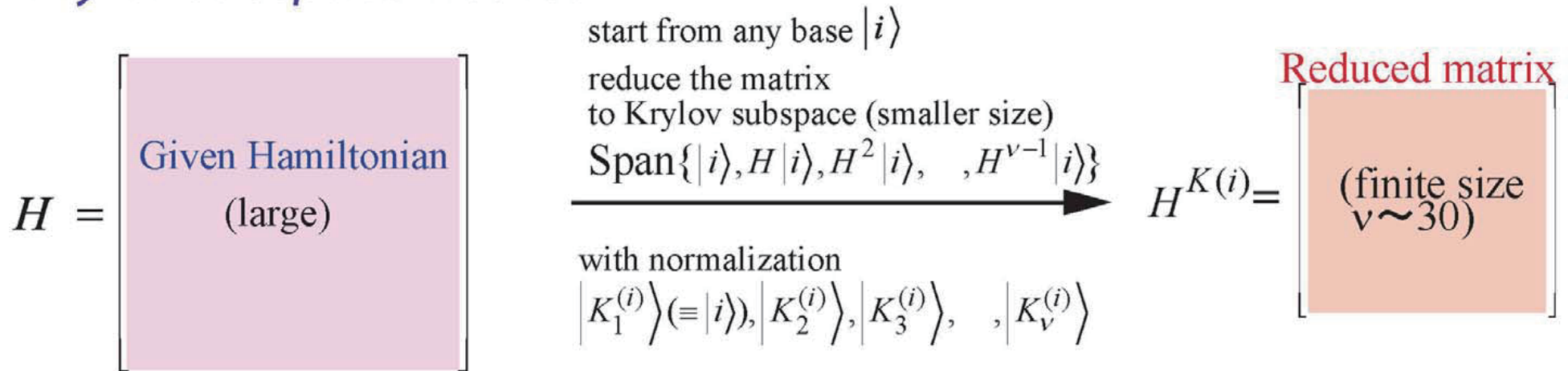
$$\Omega = H - \mu \quad (\mu > \varepsilon_N)$$



ab initio wfn. in Si crystal
 (Marzari & Vanderbilt '97)

Krylov Subspace

Krylov Subspace Method



(1) Subspace diagonalization method

(2) Shifted COCG method

Subspace Diagonalization

Takayama, Hoshi and Fujiwara
J. Phys. Soc. Jpn 73, 1519 (2004)

Diagonalize $H^{K(i)}$: $H^{K(i)} |w_\alpha\rangle = \varepsilon_\alpha |w_\alpha\rangle$

Density matrix operator: $\hat{\rho}^{K(i)} \equiv \sum_\alpha^v |w_\alpha\rangle \langle w_\alpha| f\left(\frac{\varepsilon_\alpha - \mu}{kT}\right)$,

The replacement of the density matrix:

$$\langle i | \hat{\rho} | j \rangle \Rightarrow \langle i | \hat{\rho}^{K(i)} | j \rangle$$

$$\langle i | \hat{\rho}^{K(i)} | j \rangle = \sum_n^v \langle i | \hat{\rho}^{K(i)} | K_n^{(i)} \rangle \langle K_n^{(i)} | j \rangle$$

The convergence of this summation validate the present method.

$|i\rangle$: an element of the KS

$|j\rangle$: not necessarily in the KS

The advantages of using eigenstates of $H^{K(i)}$ are

- exact calc. is available within reduced Hilbert space
- energy integration
 —————> reduced matrix size v

Error can be monitored.

$$\langle i | r_v^{(j)} \rangle = \langle i | I - (z - H) G_v^{(i)}(z) | j \rangle$$

Takayama, Hoshi and Fujiwara: Phys. Rev. B **73**, 165108 (2006)

$$(z - H)|x_n^{(j)}\rangle = |j\rangle$$

$$(z_{\text{ref}} + \sigma - H)|x_n^{\sigma(j)}(z_{\text{ref}} + \sigma)\rangle = |j\rangle$$

Green's Function

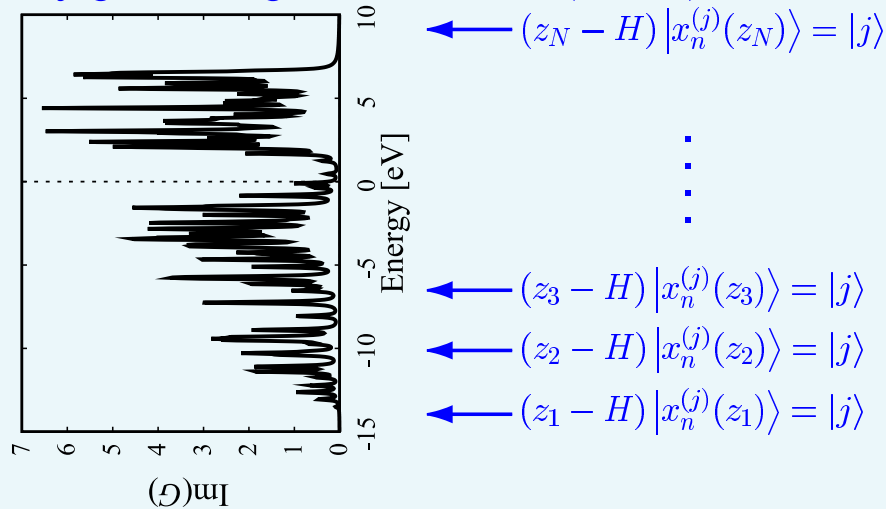
$$G_{ij}(z_k) \equiv \langle i|x^{(j)}(z_k)\rangle = \langle i|(z_k - H)^{-1}|j\rangle$$

$$\rho_{ij} = -\frac{1}{\pi} \int_{-\infty}^{\infty} \text{Im}G_{ij}(z) f\left(\frac{\varepsilon - \mu}{k_B T}\right) d\varepsilon$$

Krylov subspace by shifted z is identical to the original subspace.

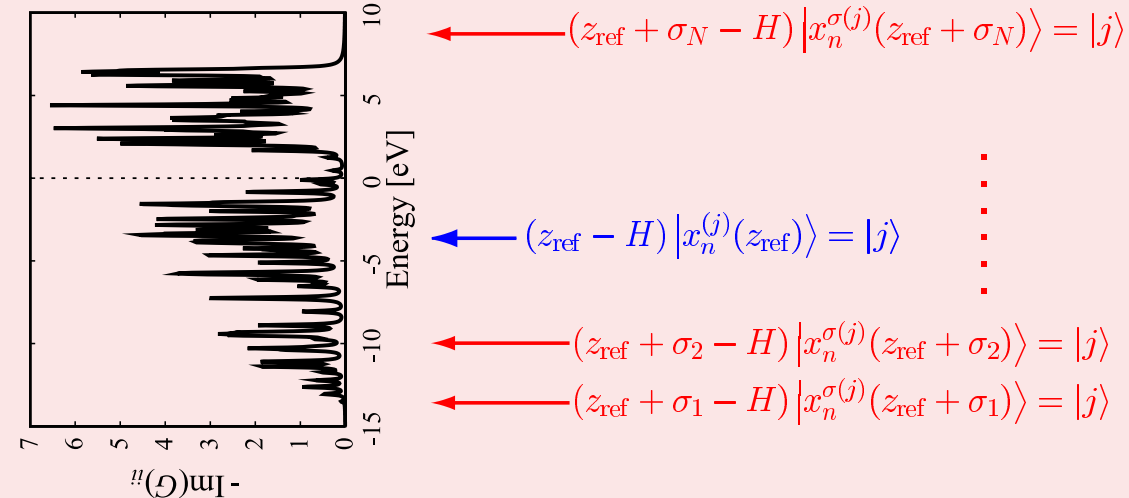
vector-matrix product: **N** times/iteration

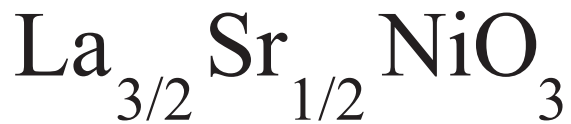
Conjugate Orthogonal CG method (COCG)



vector-matrix product: **Once** /iteration

shifted COCG method





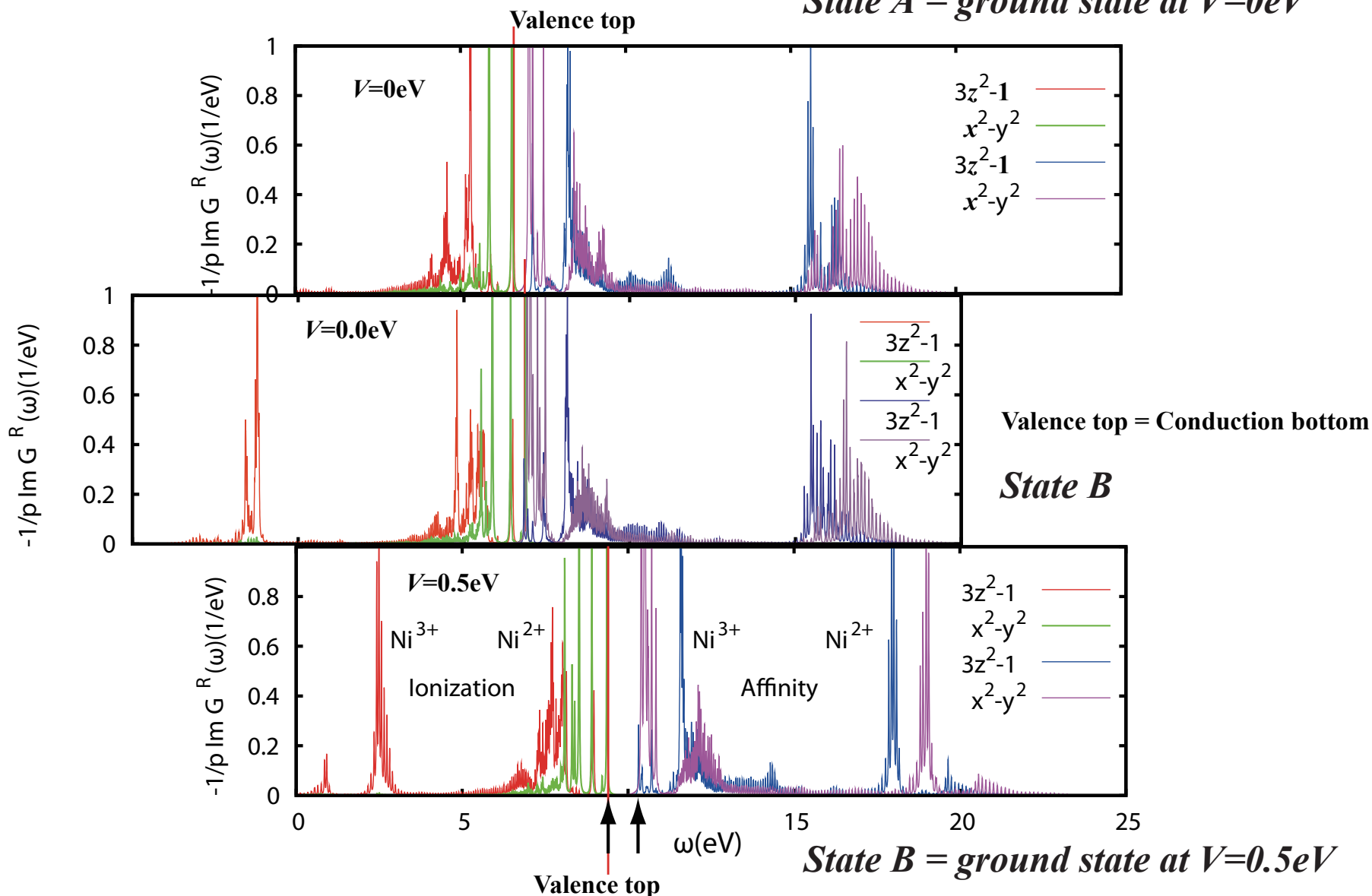
Application to Many-Body Problem

Stripe order of charge and spin

S.Yamamoto and T.Fujiwara, unpublished

$$H = \sum_{i,j,\alpha,\beta,\sigma} t_{i\alpha j\beta} c_{i\alpha\sigma}^\dagger c_{j\beta\sigma} + \sum_{i,\alpha,\sigma} \epsilon_{i\alpha} c_{i\alpha\sigma}^\dagger c_{i\alpha\sigma} + \frac{1}{2} \sum_{i,\alpha,\beta,\gamma,\delta,\sigma,\sigma'} U_{\alpha\beta\gamma\delta} c_{i\alpha\sigma}^\dagger c_{i\beta\sigma'}^\dagger c_{i\delta\sigma'} c_{i\gamma\sigma} + \frac{V}{2} \sum_{\langle i,j \rangle, \alpha, \beta, \sigma, \sigma'} c_{i\alpha\sigma}^\dagger c_{i\alpha\sigma} c_{j\beta\sigma'}^\dagger c_{j\beta\sigma'}$$

State A = ground state at $V=0\text{eV}$



Hybrid Method

$$\rho = \rho_A + \rho_B \quad \rho_A \rho_B = 0$$

$$\begin{aligned} H^{(A)} &= H + 2\eta\rho_B - (H\rho_B + \rho_B H) \\ &\cong H + 2(\eta - \varepsilon_B)\rho_B \end{aligned}$$

$$H^{(A)}\rho_A - \rho_A H^{(A)} = 0$$

Division of the Hilbert space (not division of real space)

Large-scale Electronic Structure Calculation

10^7 atoms !!
 $\sim(70 \text{ nm})^3$ system !!

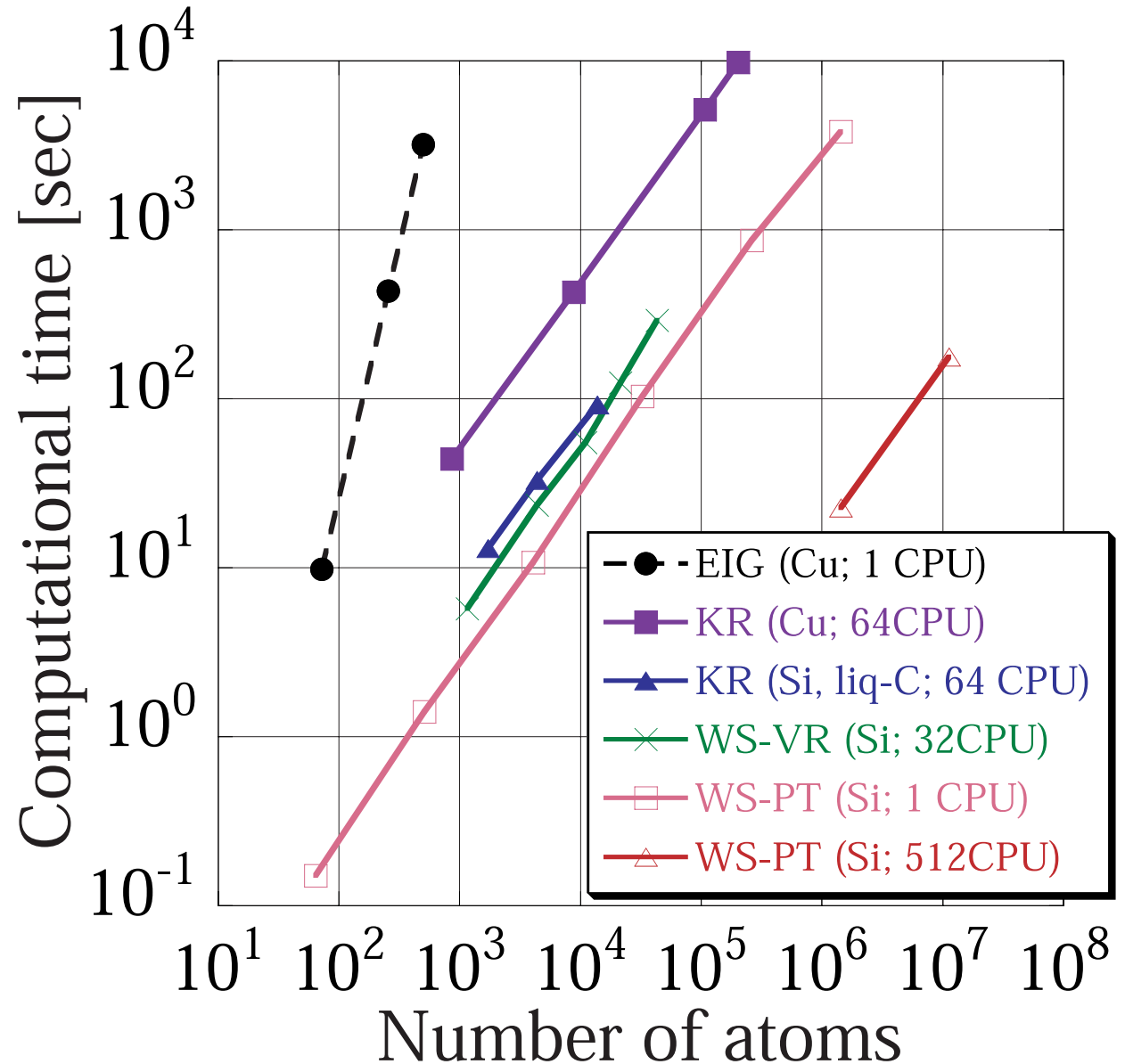
Density Matrix Formalism

Wannier state representation

Krylov subspace diagonalization

shifted COCG method

Hybrid method and parallelization



Applications

Fracture Propagation in Si Crystals
Growth Mechanism of Carbon Nanotube
Gold Helical Multishell Nanowire

Dynamical Brittle Fracture Propagation

J. Phys. Soc. Jpn **72**, 2429 (2003)

Nano-scale cleavage in Si

Phys. Rev. **B72**, 075323 (2005)

METHODOLOGY

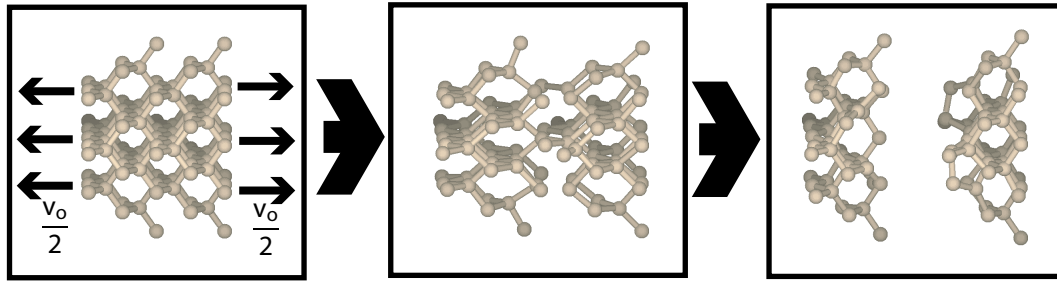
Wannier function method

J. Phys. Soc. Jpn **69**, 3773 (2000)

Krylov Subspace Diagonalization

J. Phys. Soc. Jpn **73**, 1519 (2004)

Fracture simulation (1)



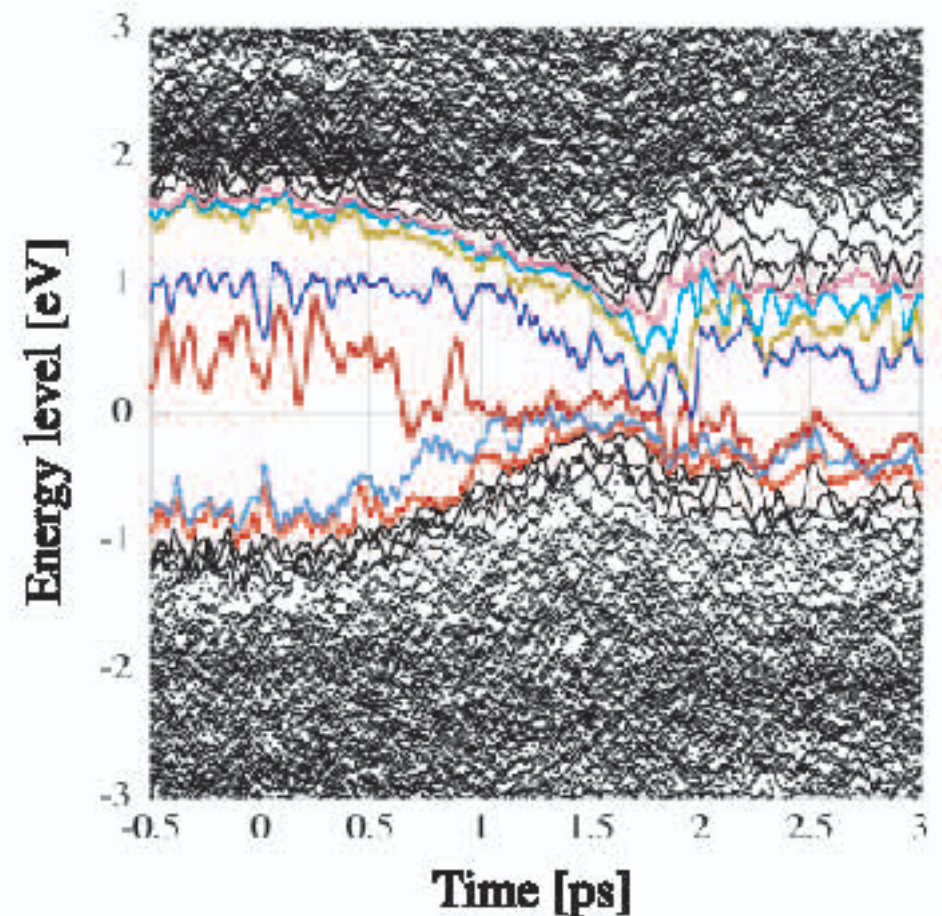
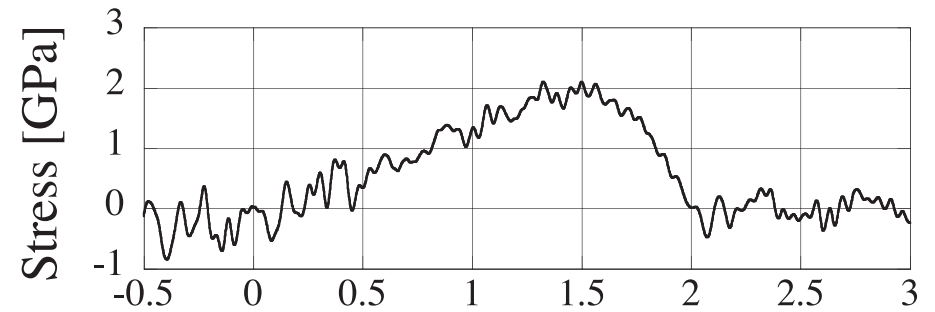
[001] external load,
isolated cluster with 91 atoms,
an initial "defect" bond,
exact diagonalization
 $v_0 = 0.1$ [K m/s]
Tight-binding, $T = 300$ K

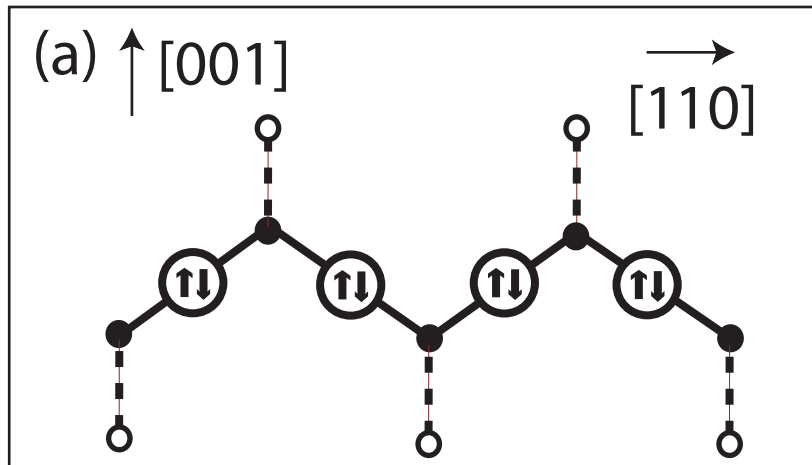
Reasonable crack-propagation speed

$$v_{\text{crack}} \approx 2 \text{ [km/s]}$$

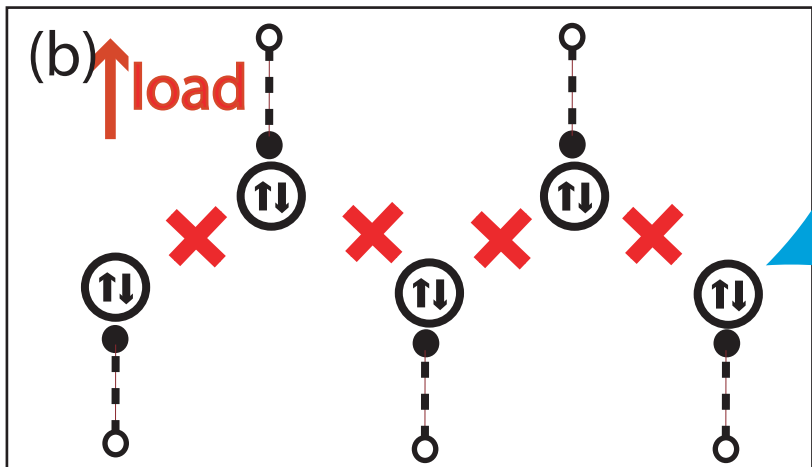
(Exp. $v_{\text{crack}} \lesssim 3.8$ [km/s])

Fracture begins as level crossings
of several "defect" states.

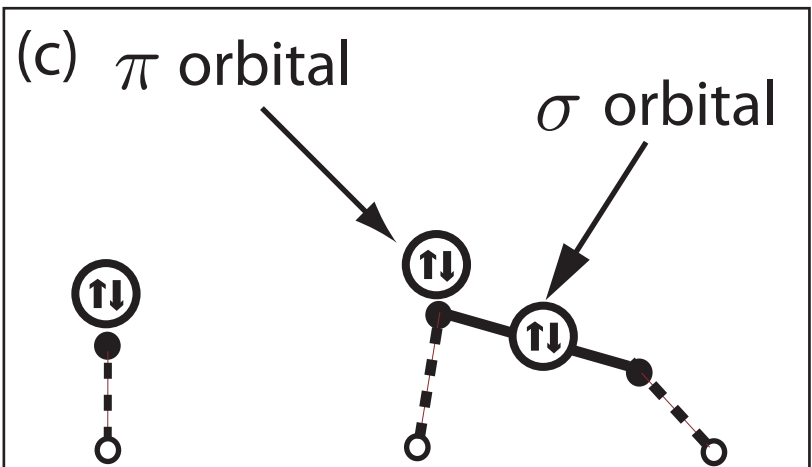




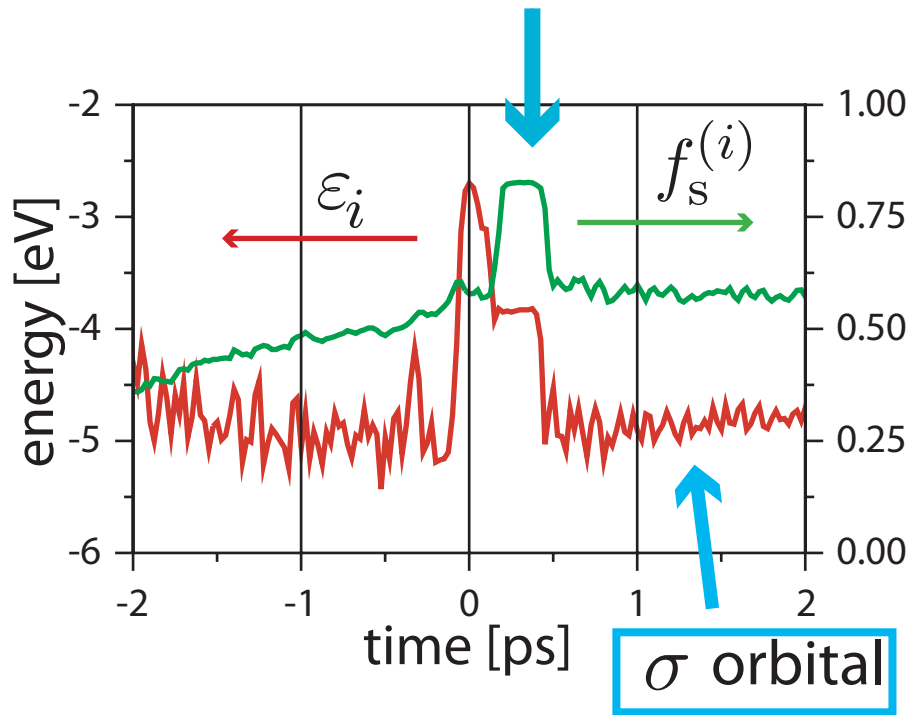
Wannier state in elementary fracture process



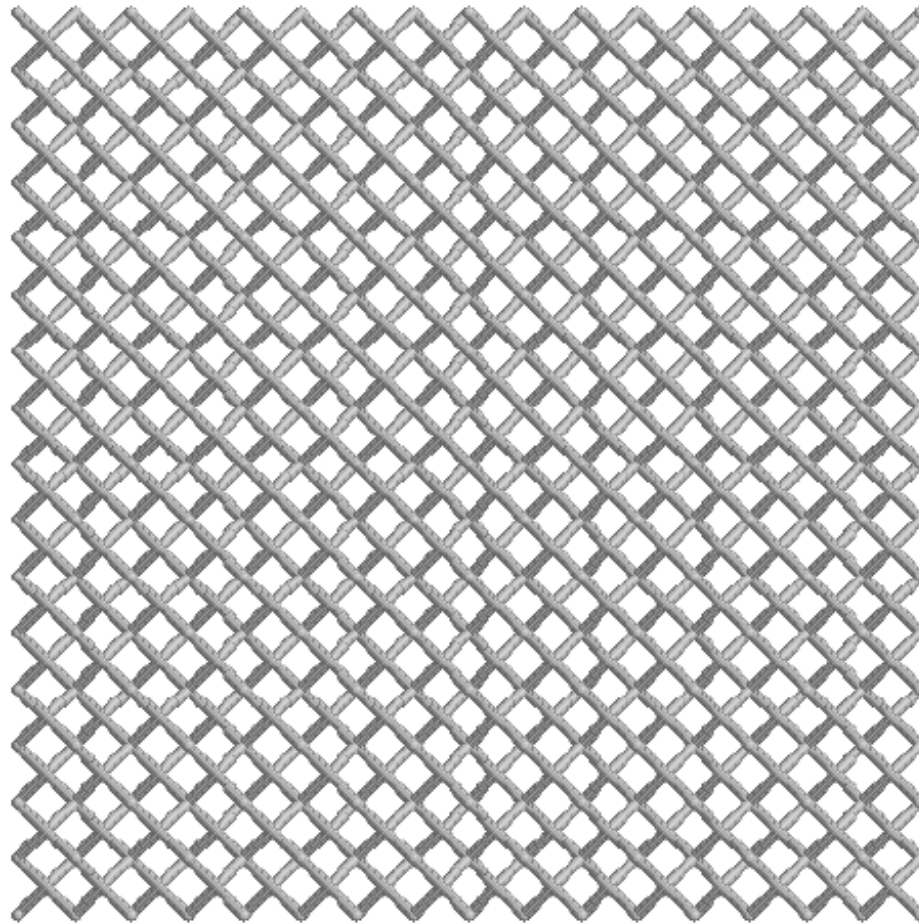
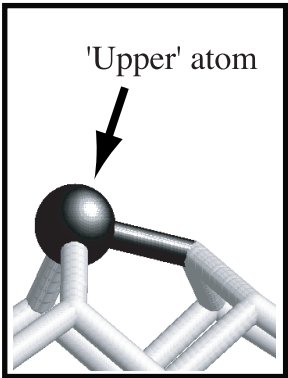
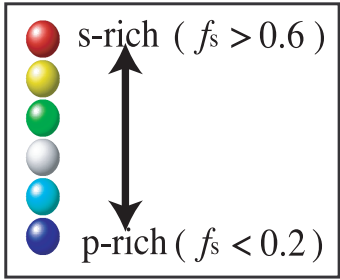
transitional atomic orbital (s-rich)



Time



4501 Si (001)



Dynamical fracture simulation in Si(001) surface (4501 atoms)

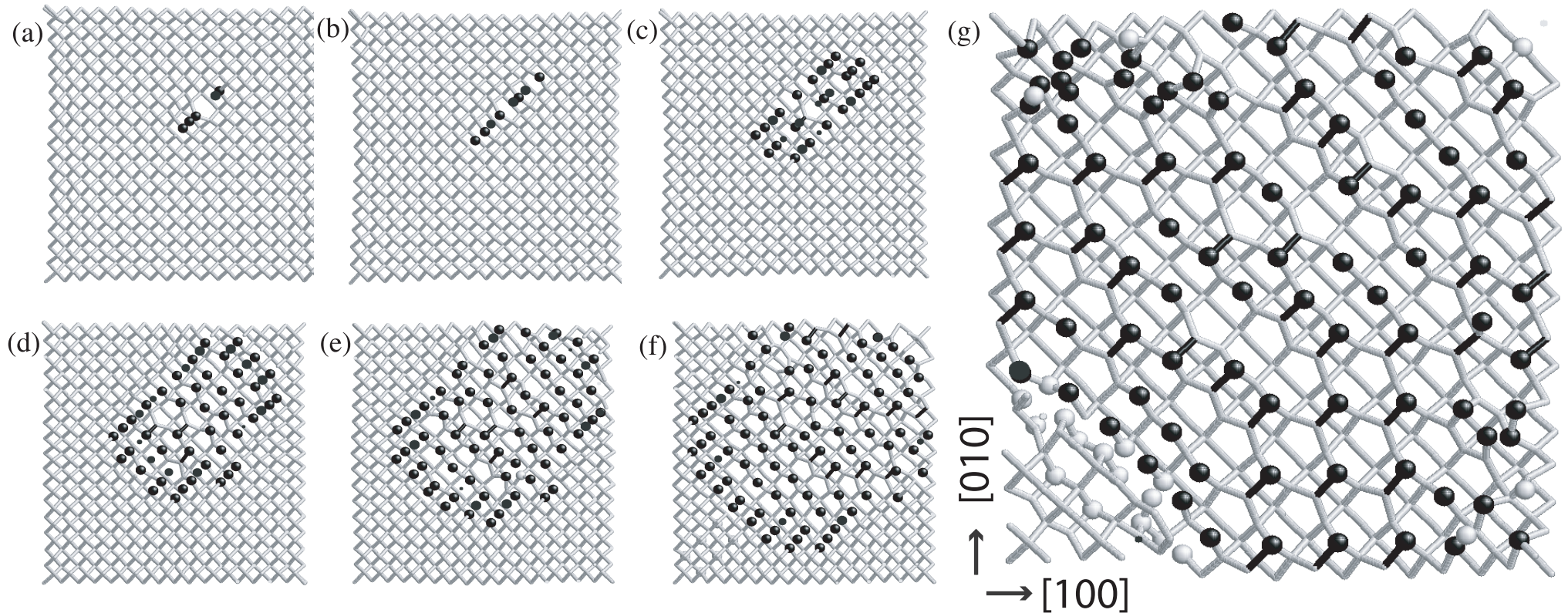
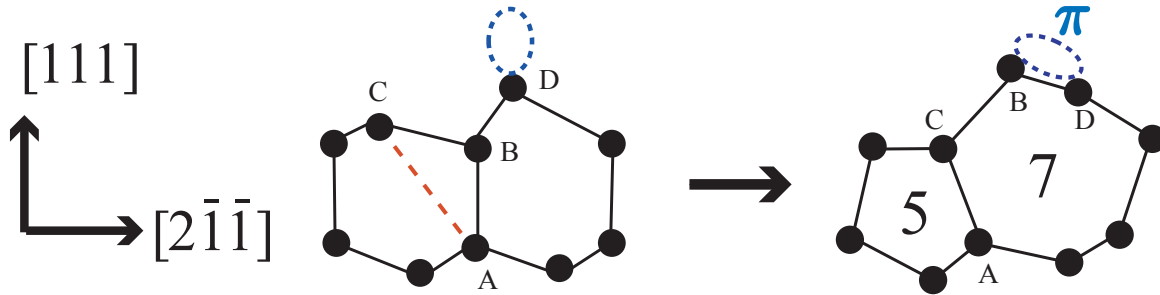


FIG. 2: Snapshots of a fracture process in the (001) plane. The sample size is $n_{100} \times n_{010} \times n_{001} = 33 \times 33 \times 33$ (4501 atoms). The time interval between two successive snapshots is 0.3 ps, except that between (f) and (g) (about 1.3 ps). A set of connected black rod and black ball corresponds to an asymmetric dimer, as in Fig. 1(b). The left-down area has not yet fractured.

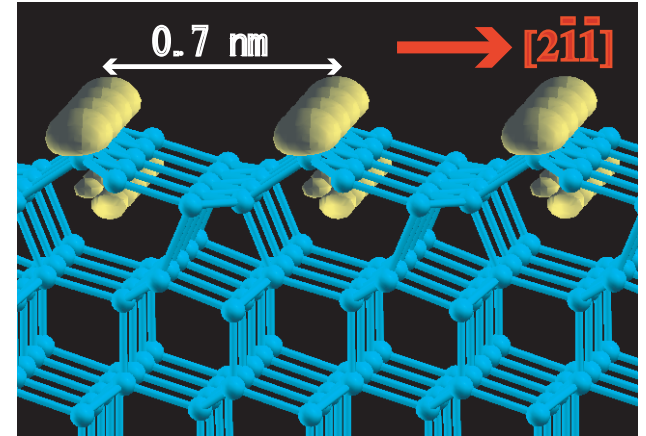
Application : Si(111)-(2x1) cleavage simulation

Hoshi, Iguchi and Fujiwara
Phys.Rev. B72, 075323 (2005)

→ π -bonded (Pandey) structure with tilting
Steps are formed in large (10-nm-scale) samples

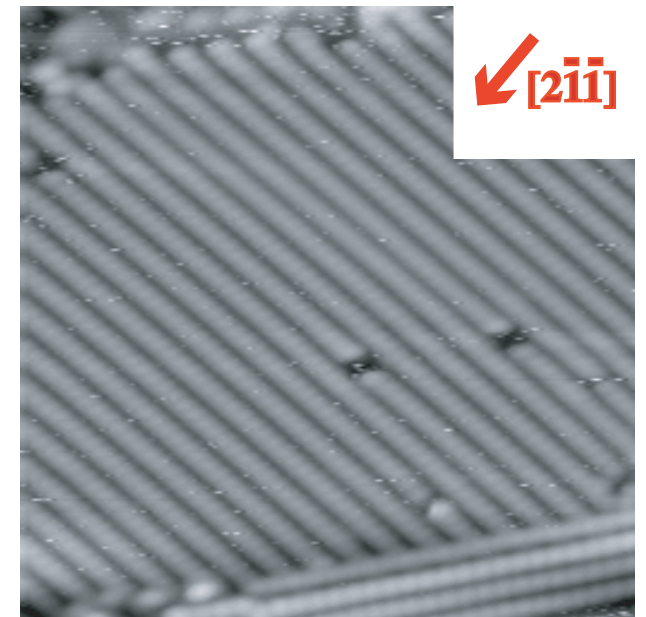


Surface (π -bonding) state

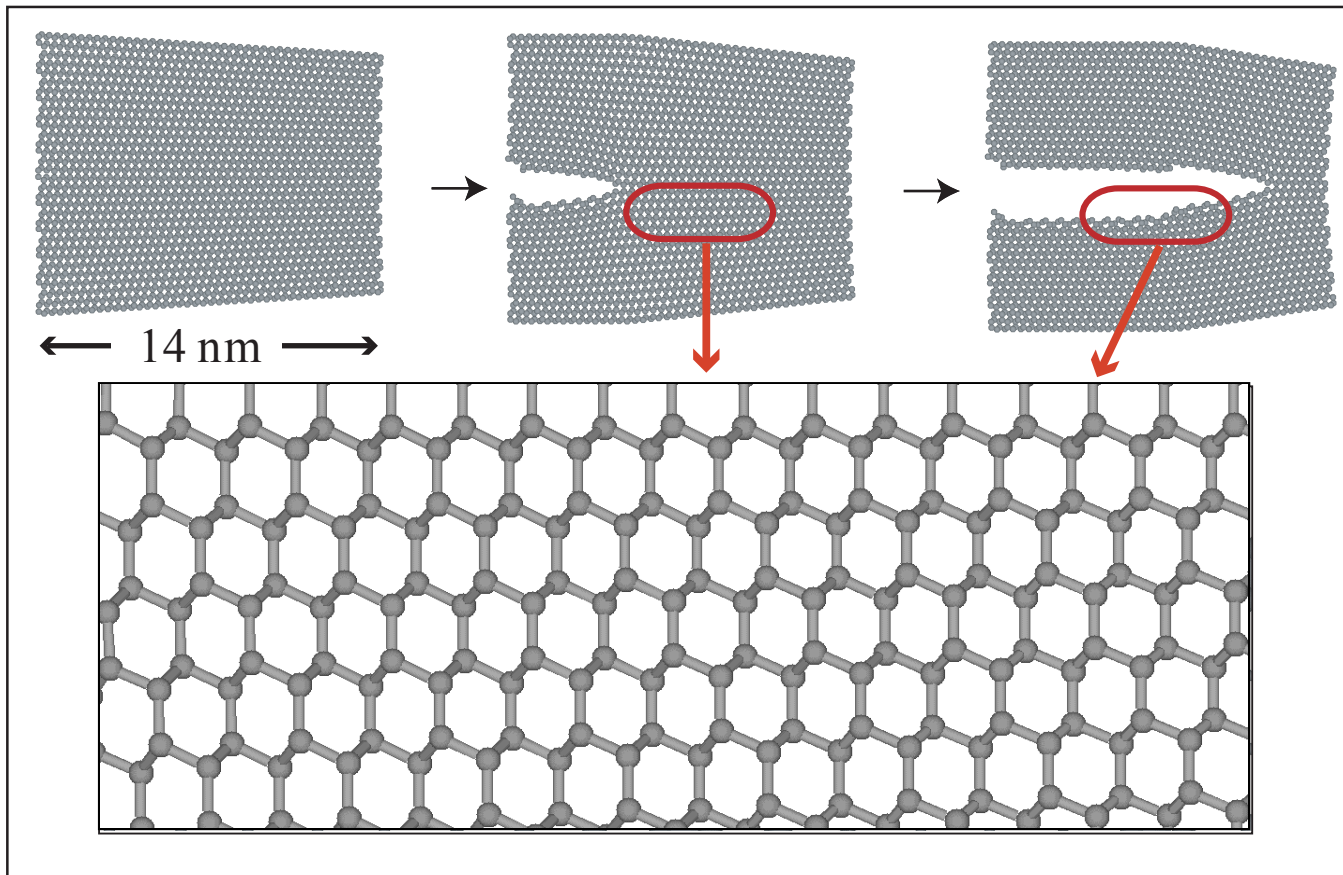


Example of STM experiment

← 15 nm →



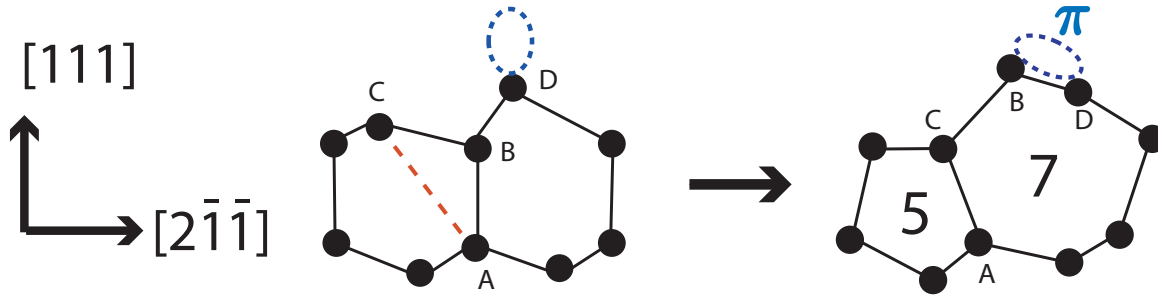
Mera, *et al.*, Ultramicroscopy, (1992)



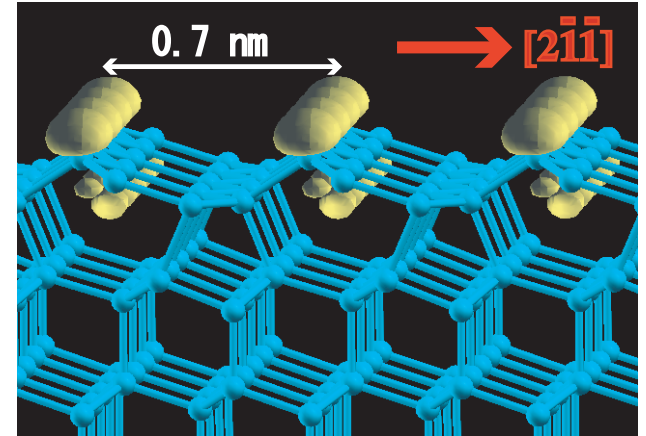
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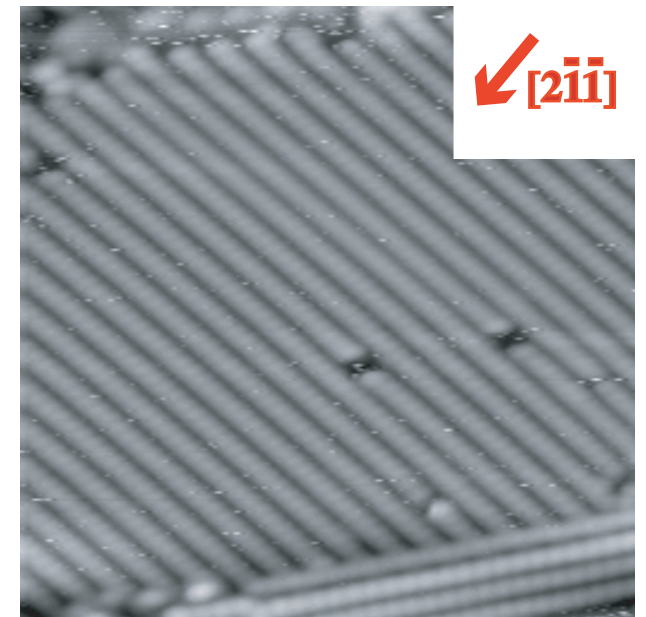


Surface (π -bonding) state

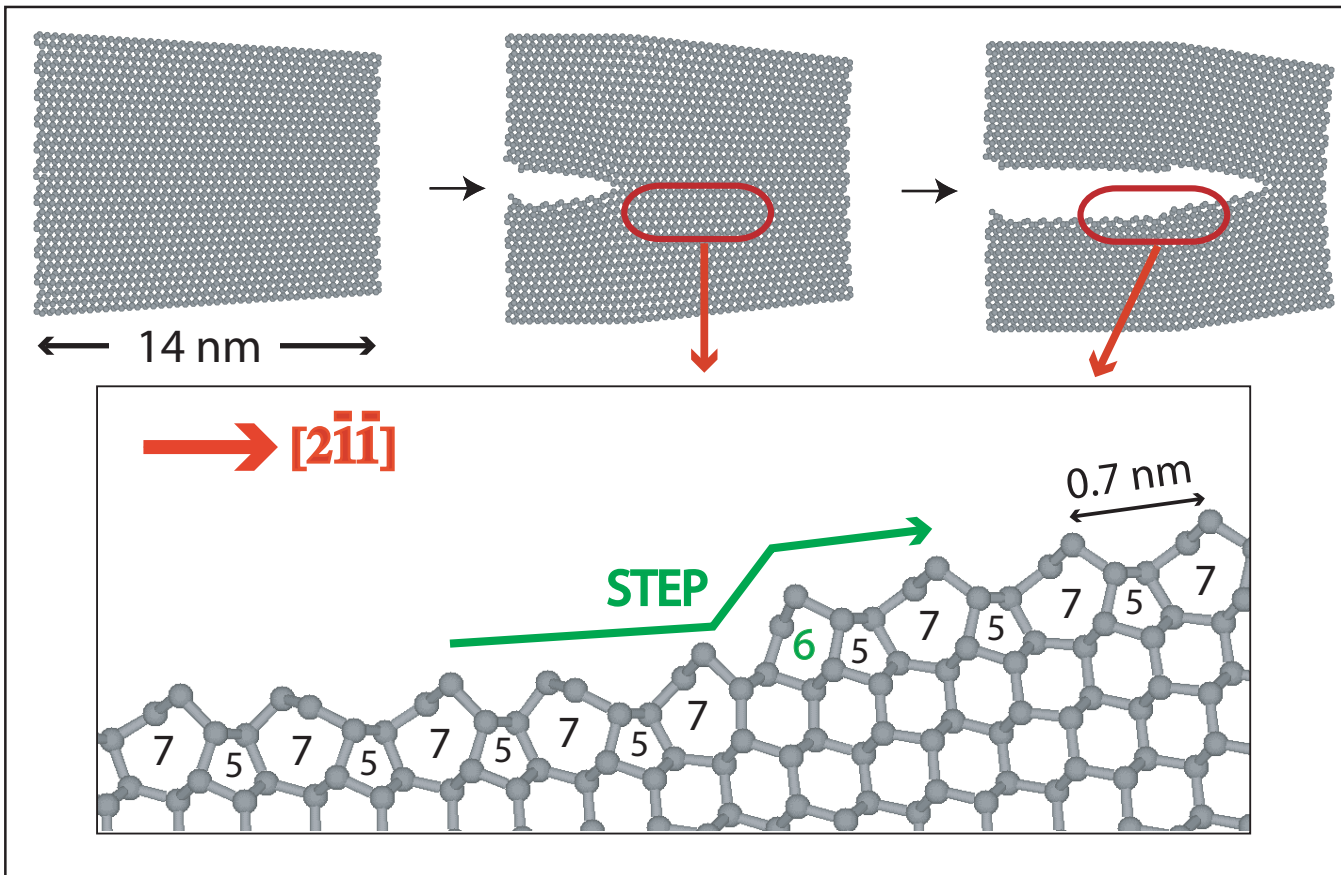


Example of STM experiment

← 15 nm →



Mera, et al., Ultramicroscopy, (1992)



Quantum mechanical analysis on Si(111) reconstruction process

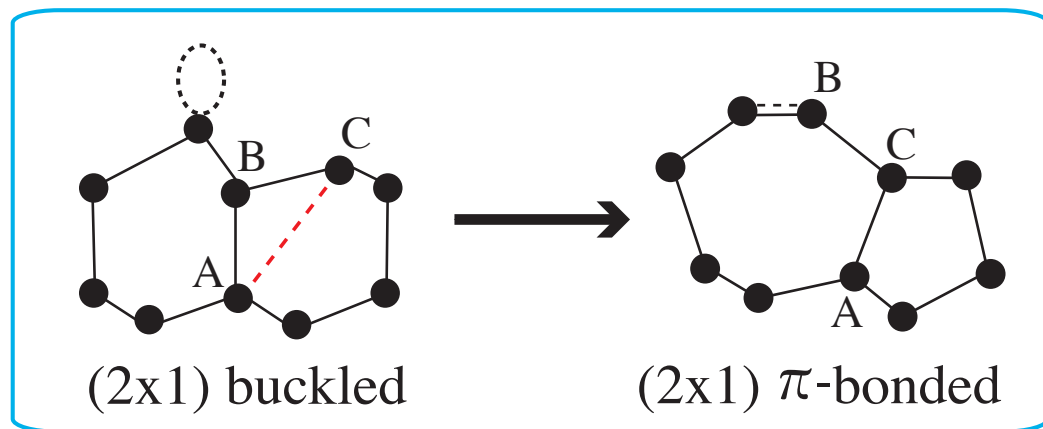
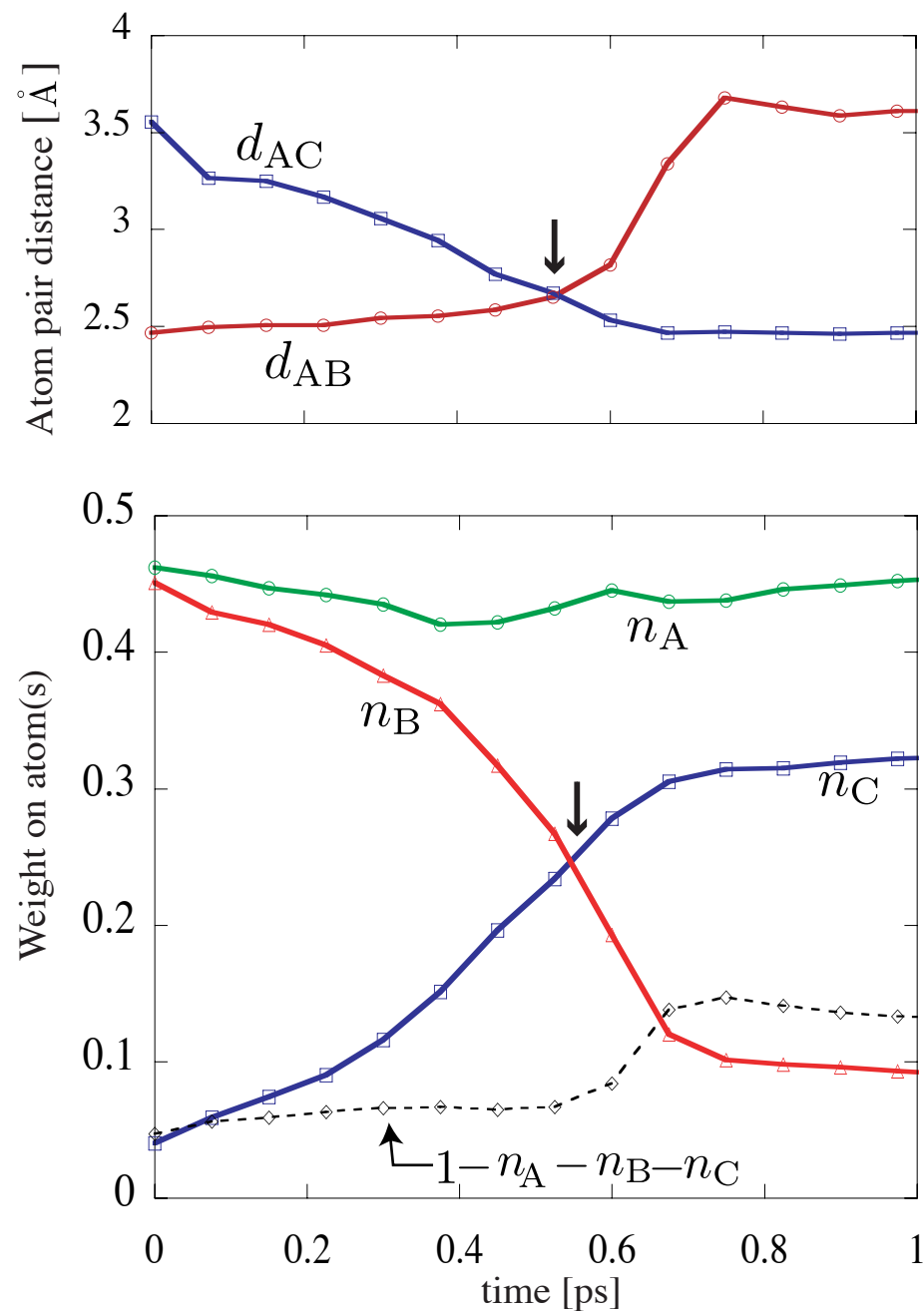
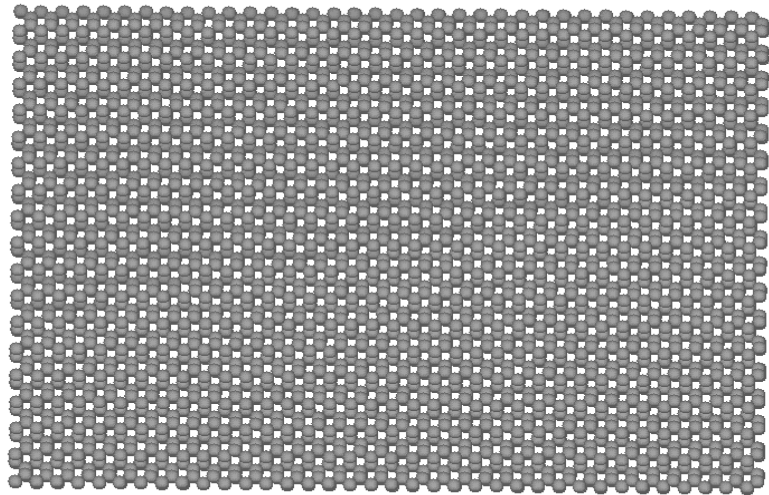


Fig. Quantum mechanical analysis of the wfn.

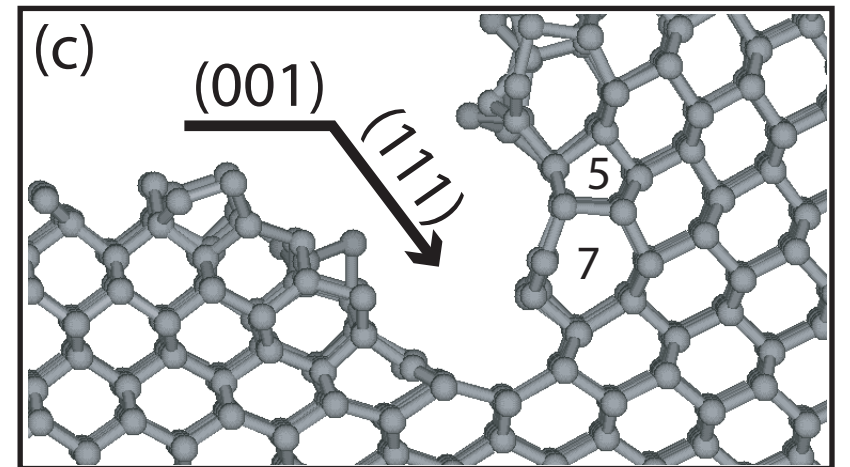
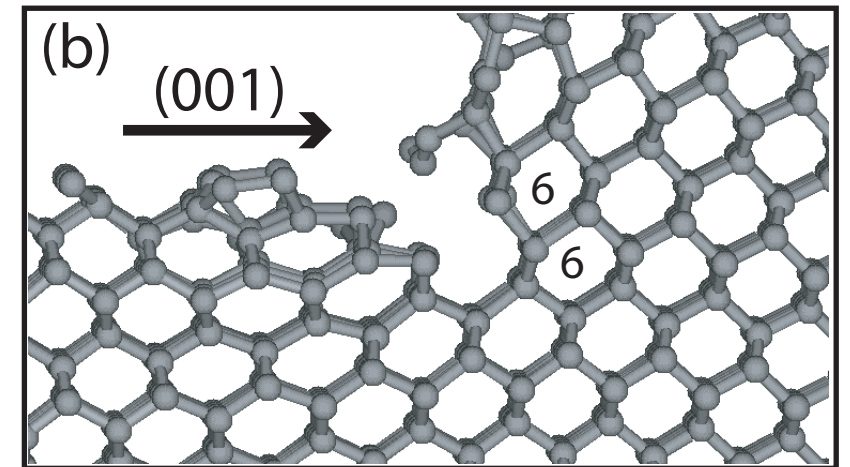
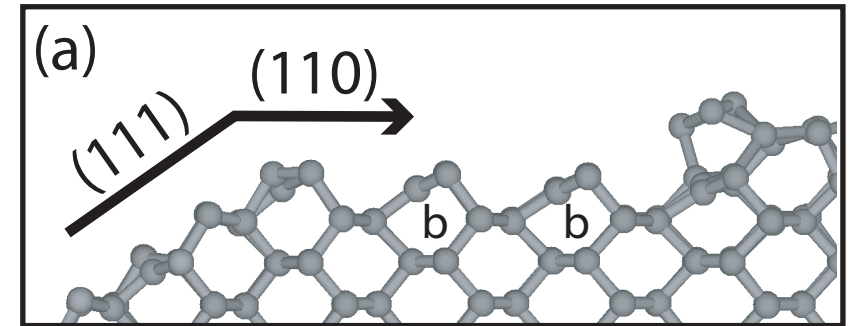
n_A, n_B, n_C : weights on the atoms



Bending of cleavage path into favorite planes

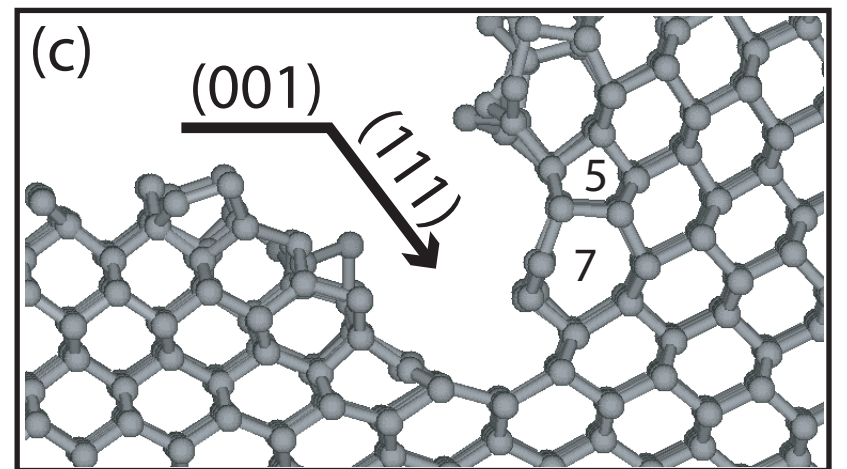
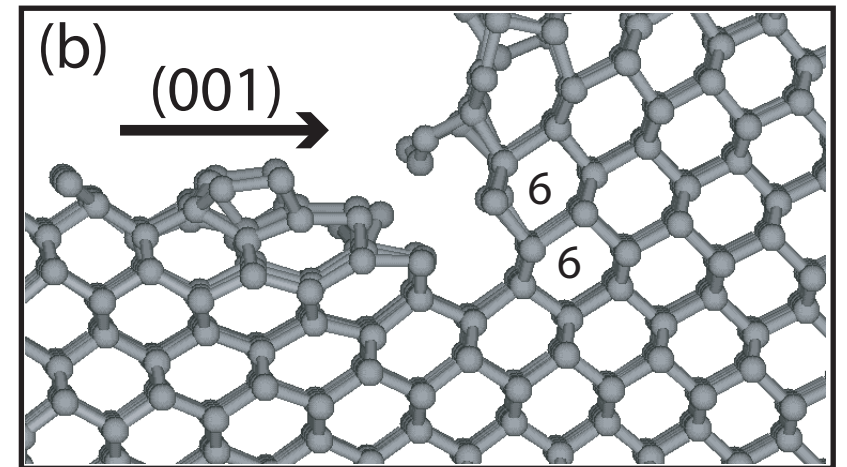
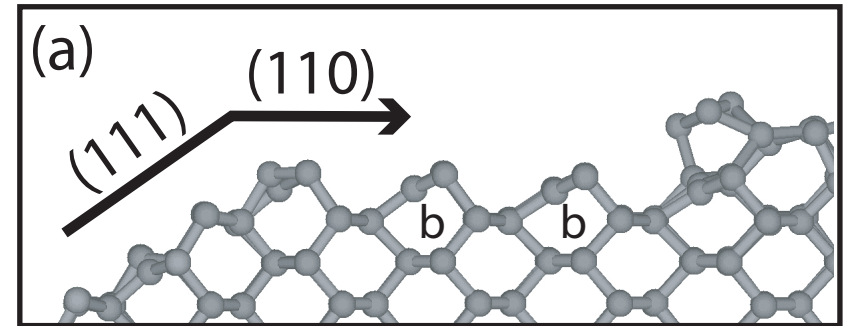
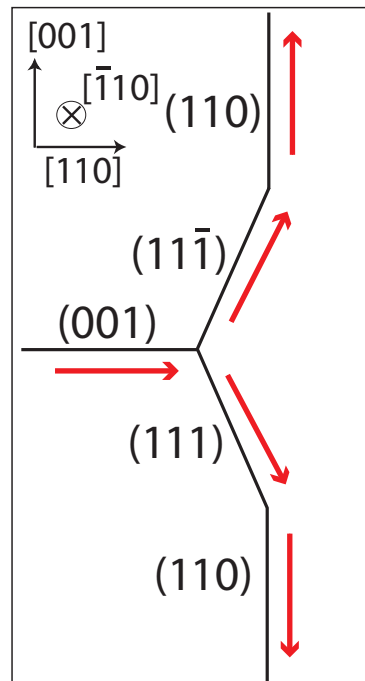
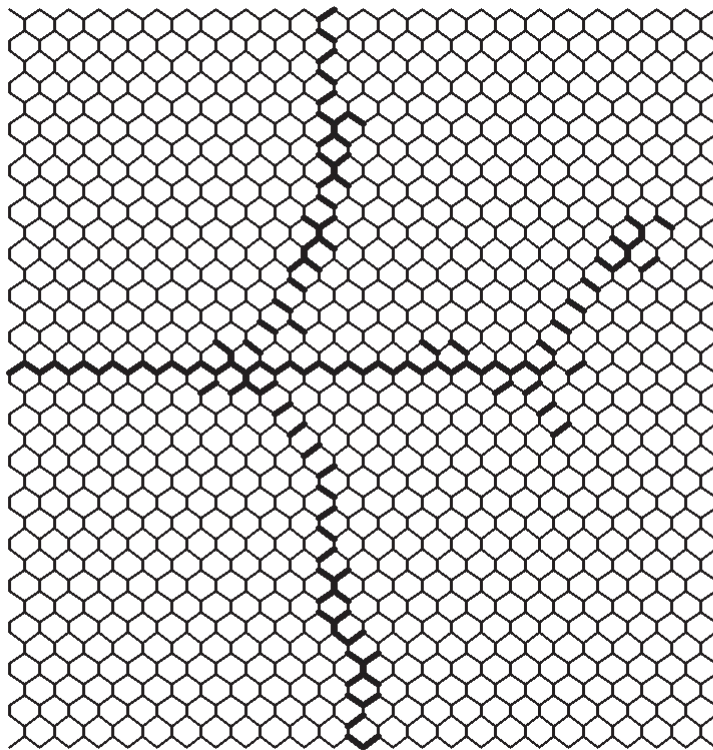


Hoshi, Iguchi and Fujiwara
Phys.Rev. B72, 075323 (2005)



Bending of cleavage path into favorite planes

Hoshi, Iguchi and Fujiwara
 Phys.Rev. B72, 075323 (2005)



Summary of Si fracture simulation 10^2 - 10^5 atoms

(001) plane

- (1) **Elementary process** for fracture : (2x1) structure
 - (a) Two-stage elementary process : dehybridization and charge transfer
 - (b) Formation of flat Si(001) surfaces with asymmetric dimers
 - (c) Two fracture mechanisms within a single atomic layer
between the [110] and [1-10] directions
- (2) **Anisotropic** behavior and **step** formation
- (3) Crossover between nanoscale and macroscale behavior

(111) plane and change of cleavaged plane

- (1) **Surface reconstruction** for fracture : (2x1) structure
 - Two-stage elementary process
 - Charged atom and buckled bonded pair
- (2) Step formation through (1-1-1) and (100) planes

Application-2

Fracture Propagation in Si Crystals
Growth Mechanism of Carbon Nanotube
Gold Helical Multishell Nanowire

Zhang, Hoshi and Fujiwara
unpublished

Elongated growth process of SWCNT from liquid carbon

(acceleration test: high density, high temperature)

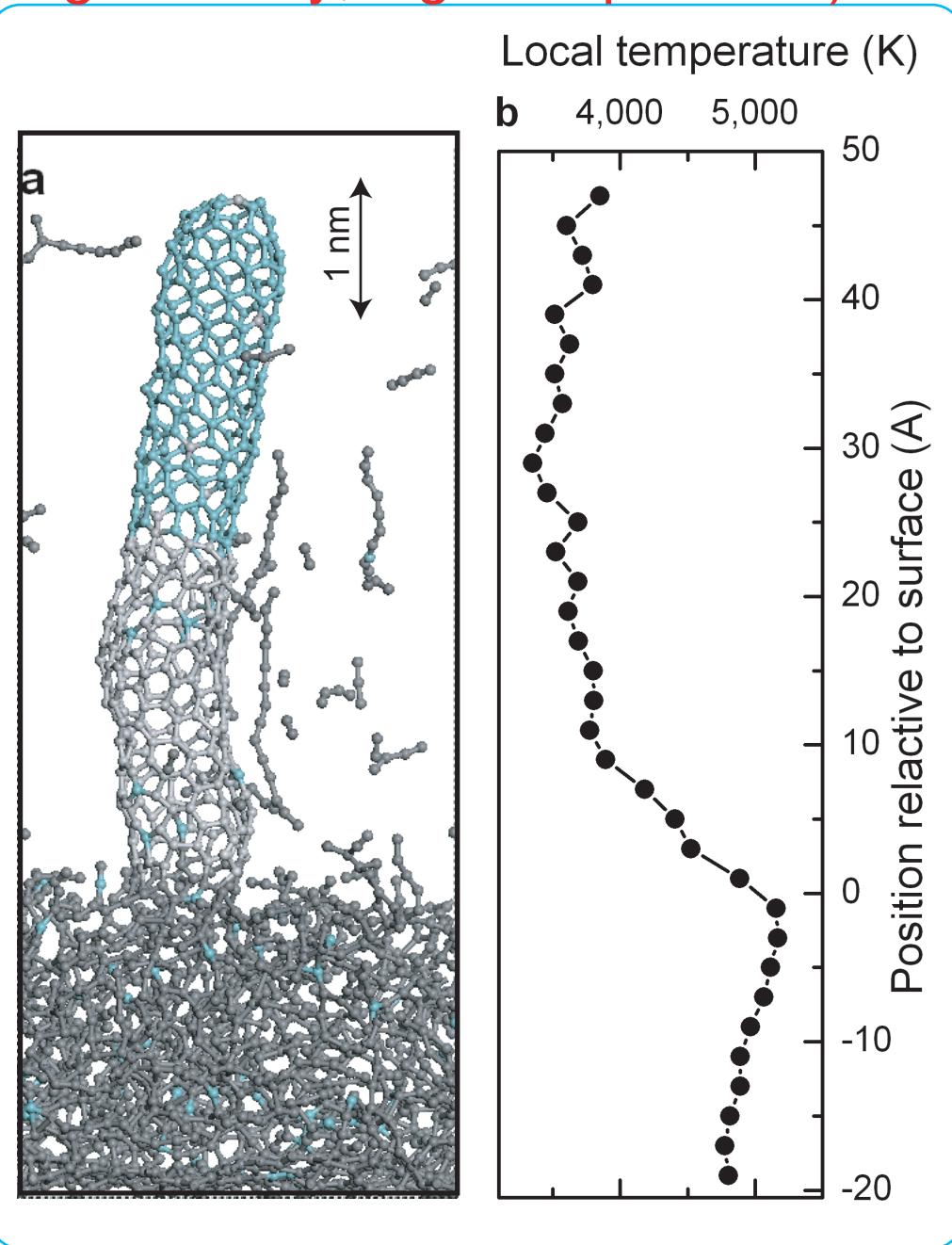
SWCNT : (9,0)
Density : 1.7 g/cm³
Growth velocity: 30 m/s
Simulation time : 86 ps

Elongated by more than 2nm
Follow structure of wall
Defects

The Nose-Hoover thermostat
is used to control the total
kinetic energy of the system.
(T=4800 K)

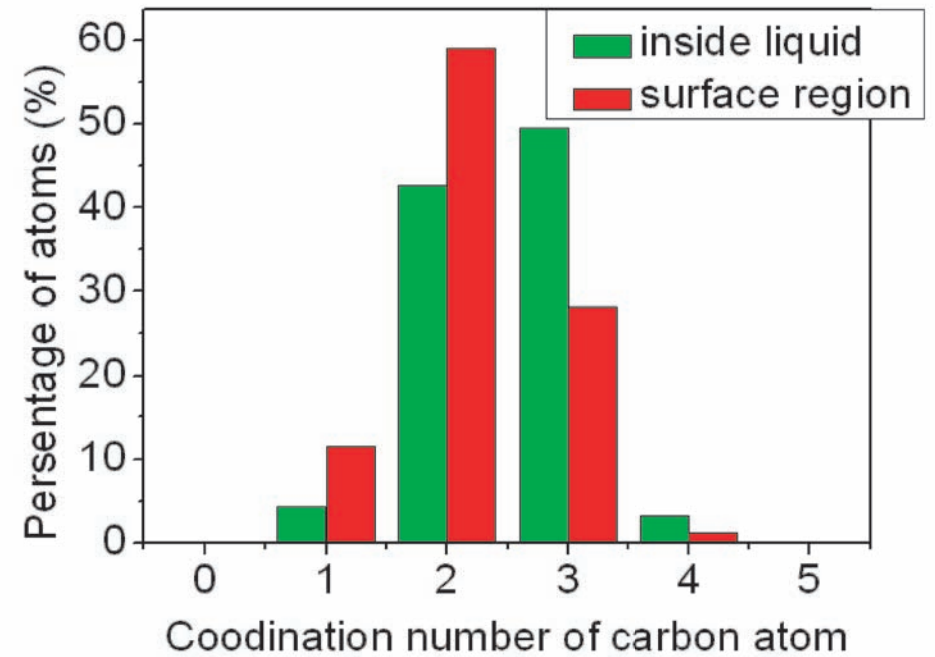
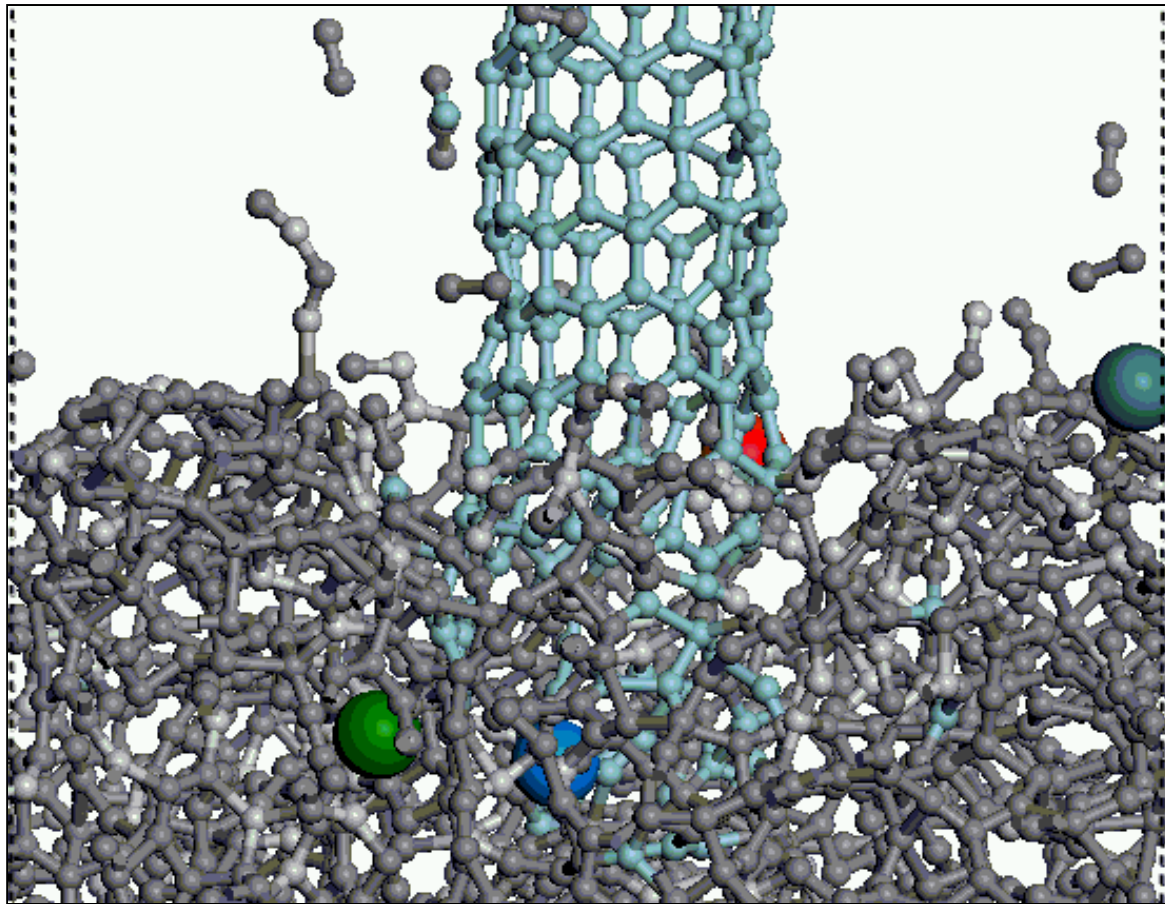
The local temperature of
the upper CNT tip is kept
to be lower than that
of the liquid part. (3800K)

Cooling process



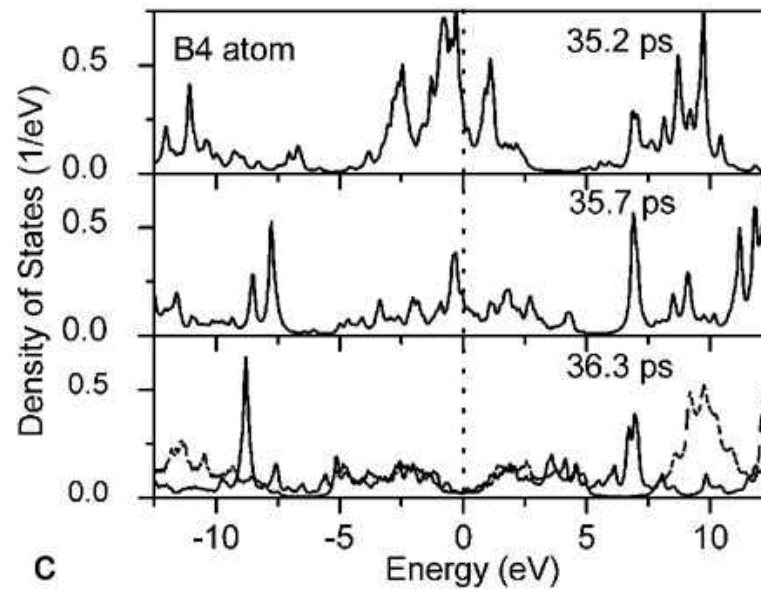
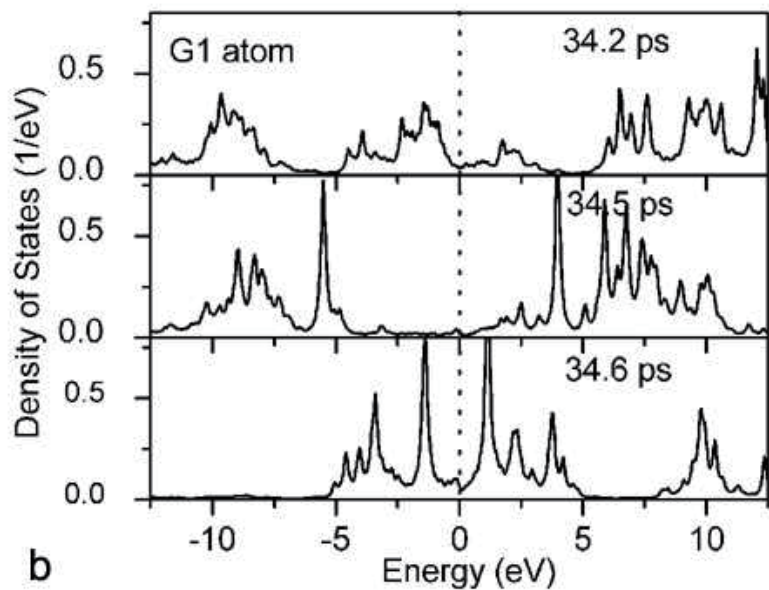
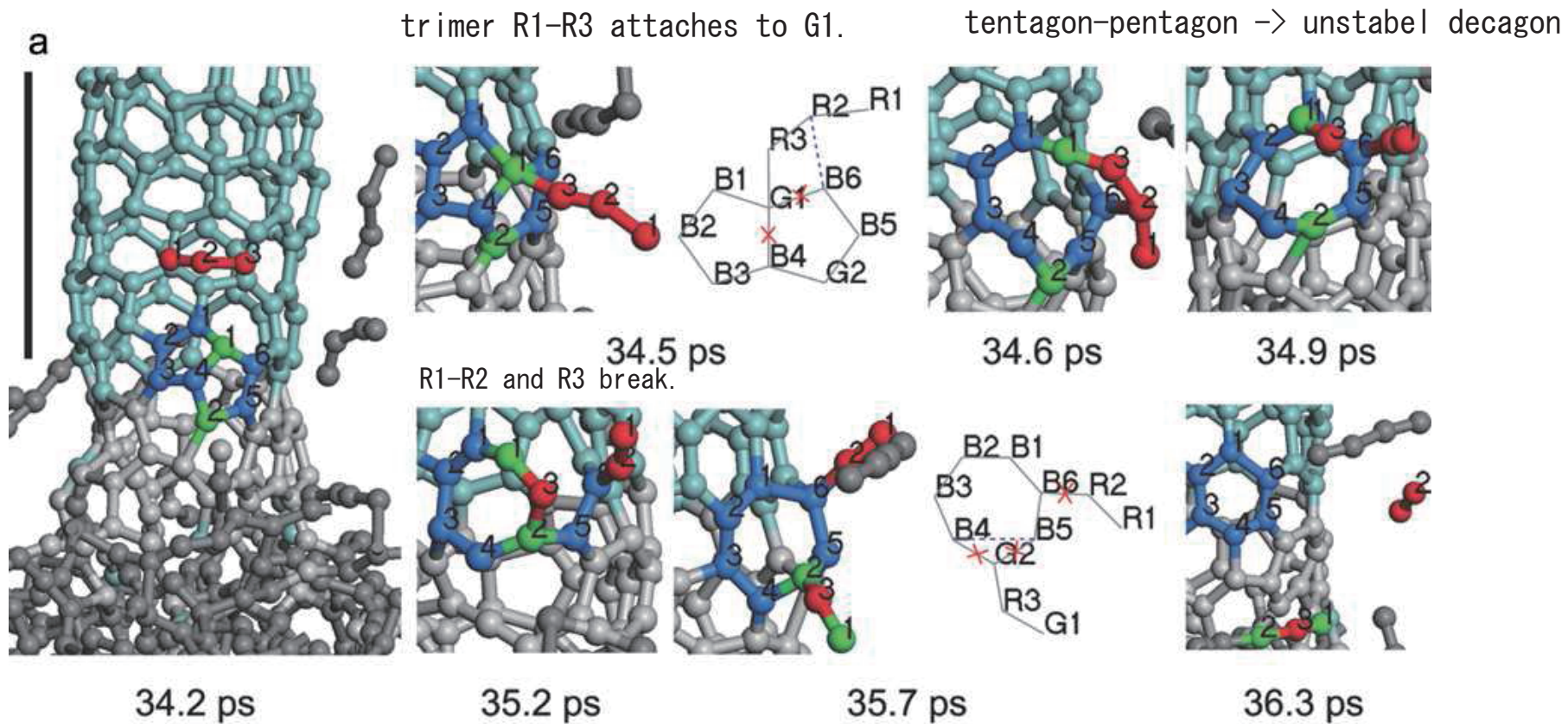
The atoms come into CNT through surface region

Two-fold carbon atoms are majority in surface region



9.0 ps → 20.1 ps

Defect Healing Process



Comparison with
the ideal (9,0) CNT

Summary of Simulation for metallic systems
: Growth mechanism of C-nanotube

Large scale Quantum Mechanical Simulation (Krylov subspace)

- (1) **Liquid precursor** for growth of C-nanotube
- (2) **Linear chain-like structure** in the surface region of liquid C
- (3) **Defect healing** process

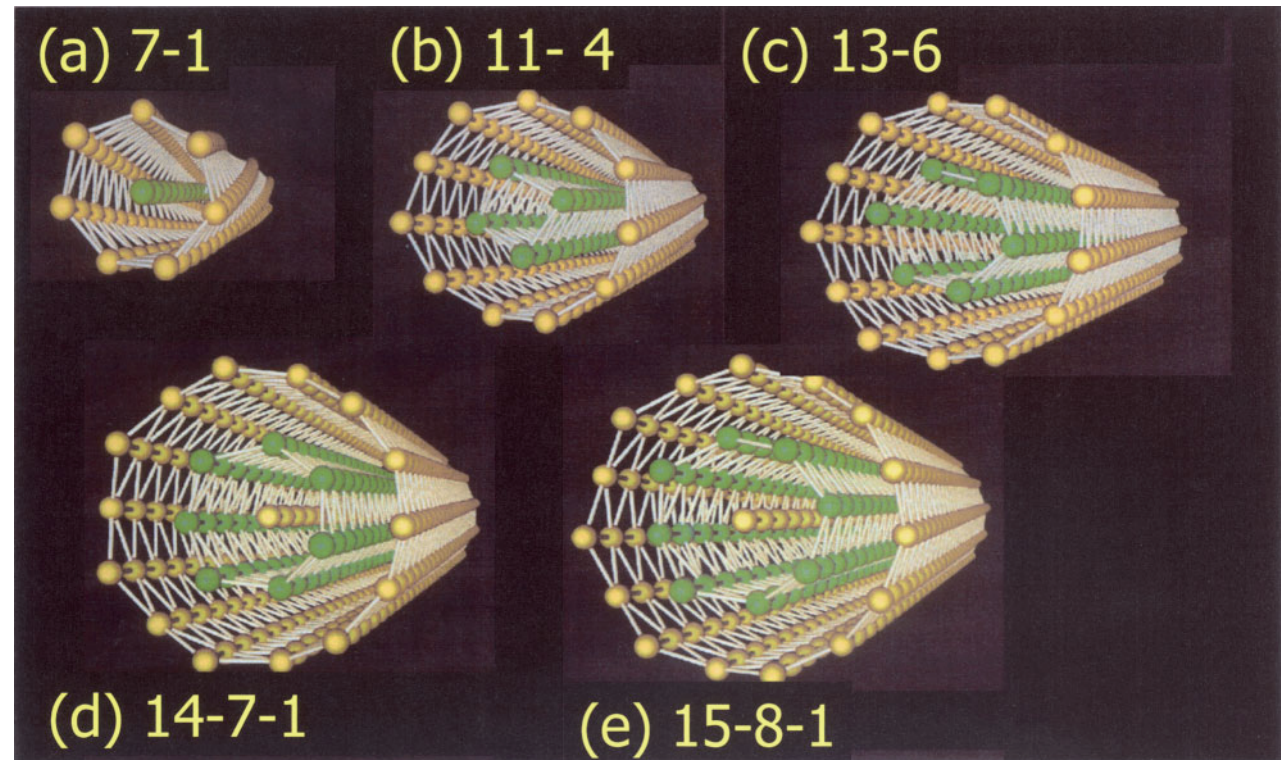
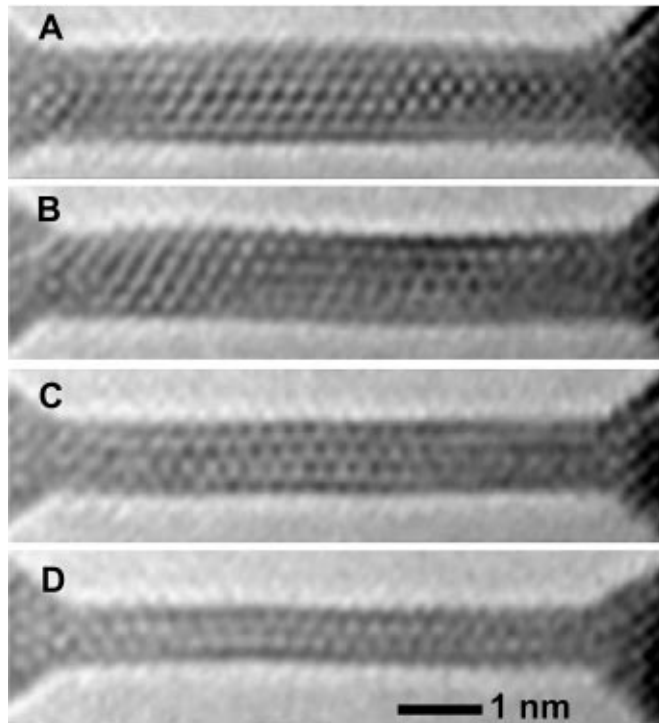
Applications

Fracture Propagation in Si Crystals
Growth Mechanism of Carbon Nanotube
Gold Helical Multishell Nanowire

Iguchi, Hoshi and Fujiwara,
to be published in PRL

Gold helical multishell nanowires

Magic number, Multi-shell structure, Helicity

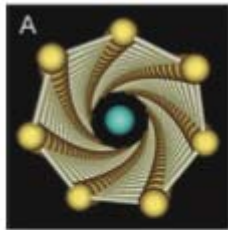


Y.Kondo and K.Takayanagi
Science **289**, 606 (2000)

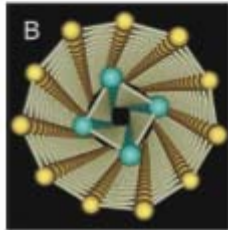
Y.Ohshima, Y.Kondo and K.Takayanagi,
J. Electron Microscopy, **52**, 49 (2003)

Magic number

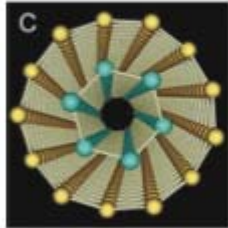
Kondo *et al.*, Science **289**, 606 (2000)



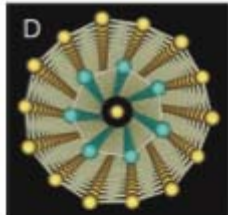
7-1



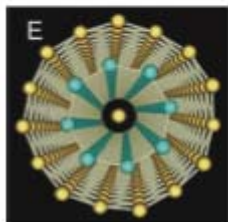
11-4



13-6

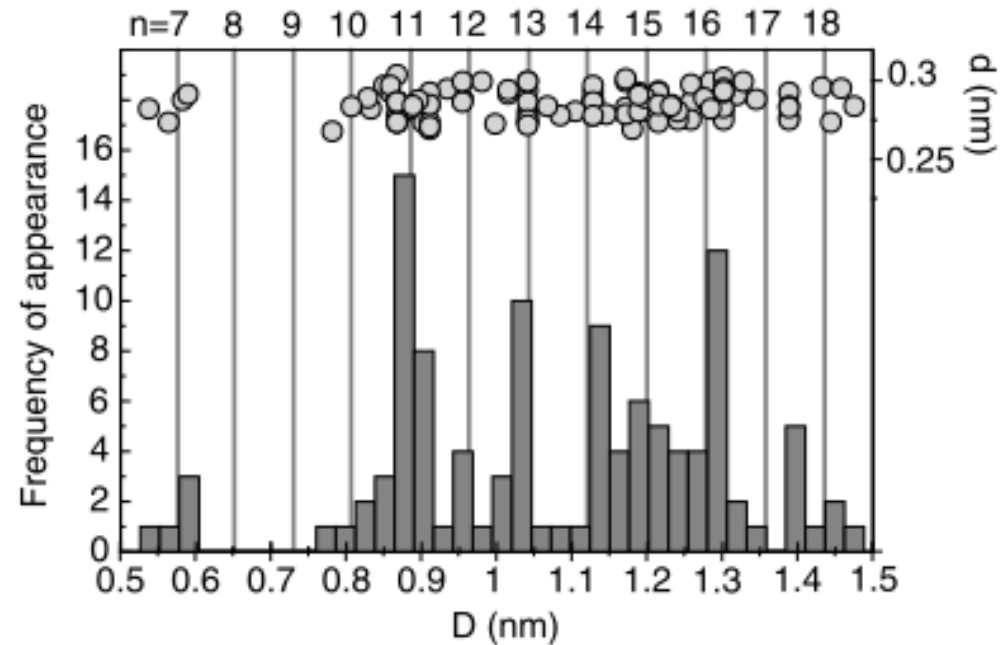


14-7-1



15-8-1

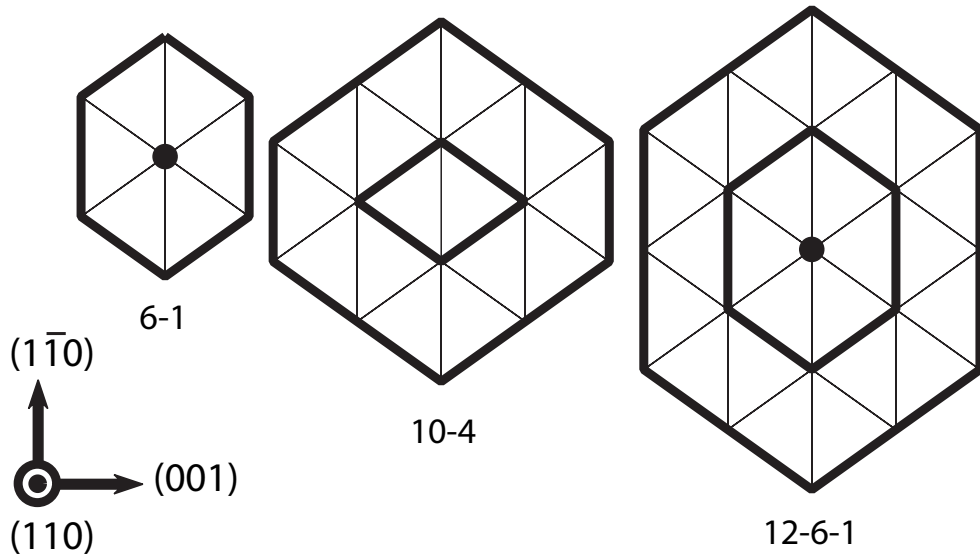
1. Peak structure in appearance frequency



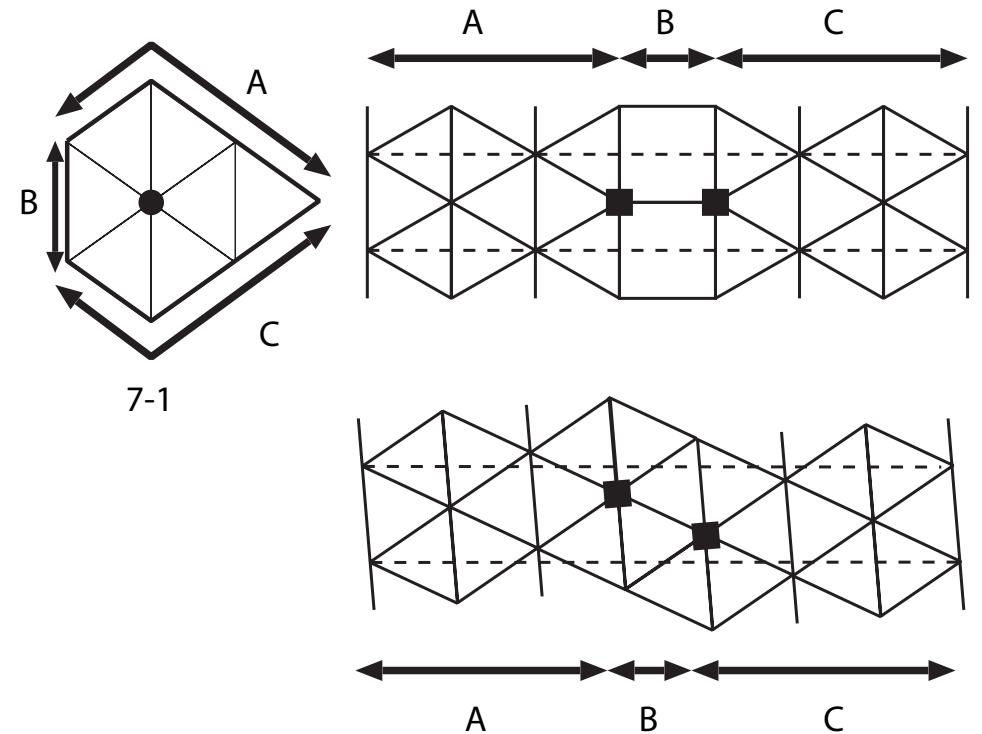
2. Difference of atom numbers between the outermost and next shells is seven.
(Magic number)

Two-stage model for helical multishell nanowire

(a) Ideal multishell structure without helicity



(b) Slip deformation of an atom row

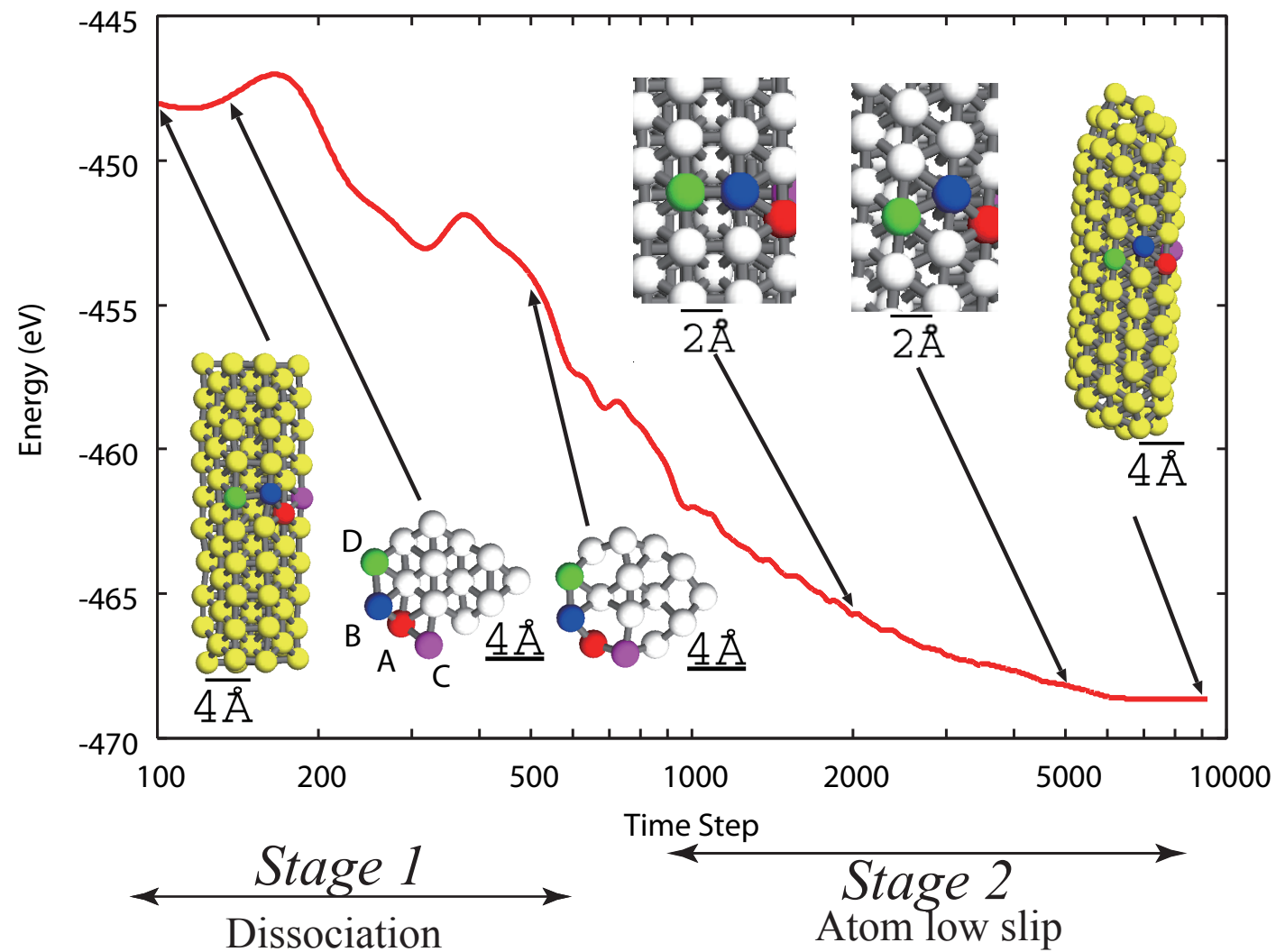


Stage 1: Dissociation of atoms on the outermost shell from the inner shell

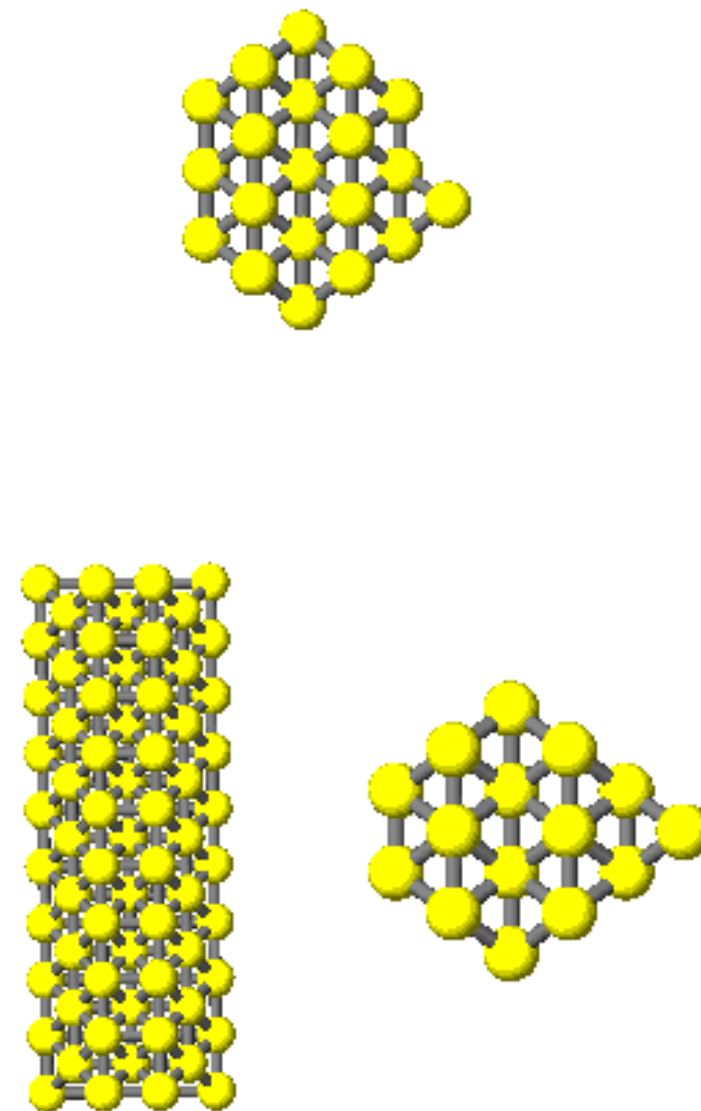
Stage 2: Slip deformation to introduce the helicity

Energy and structure change

11-4 nanowire

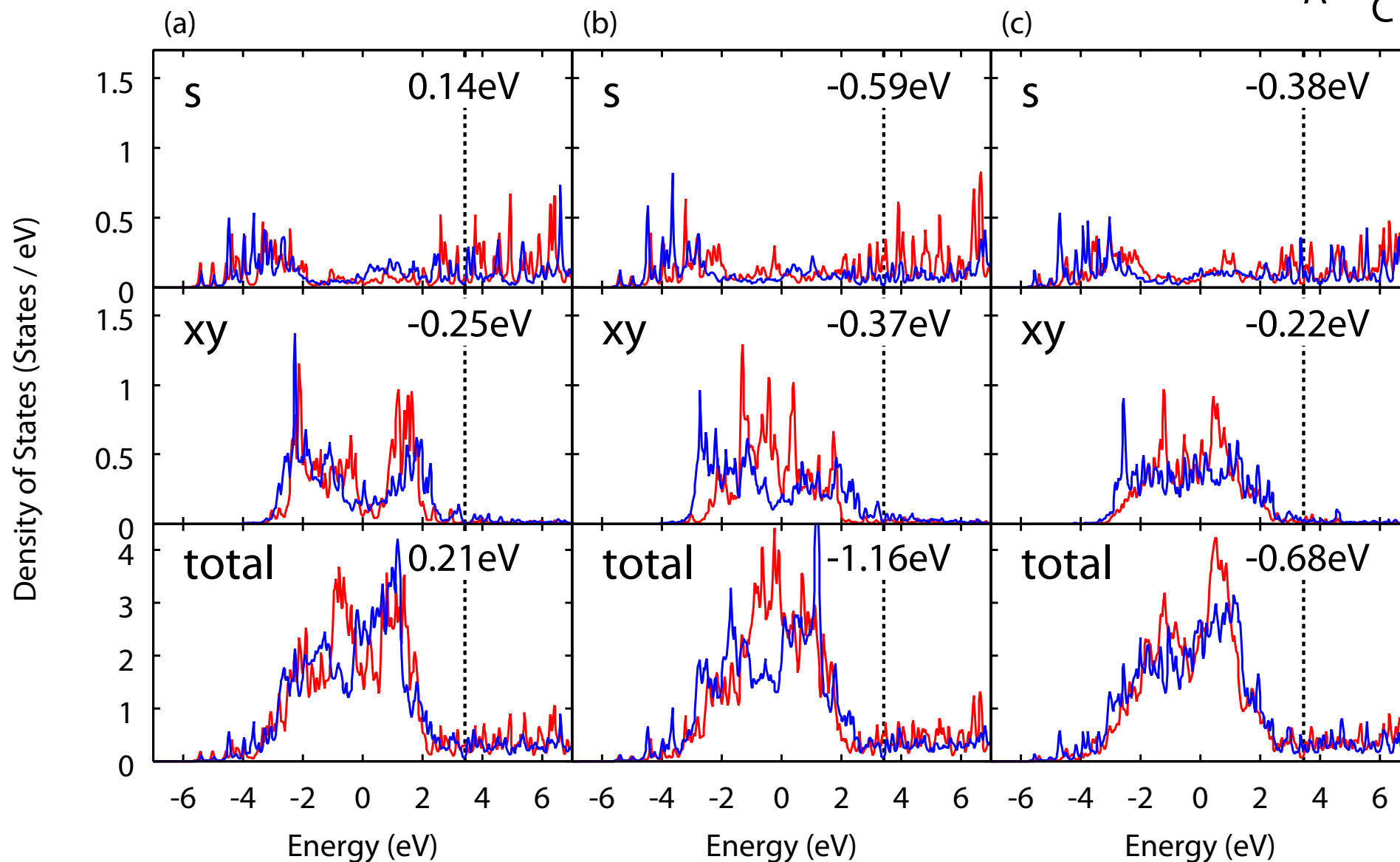
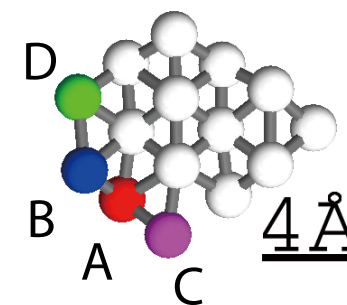


13-6-1



Gold helical multishell nanowires

Local Density of States (11-4 Au nanowire)



Atom A t=0 and 0.5 ps

Atom C at t=0 and 0.5 ps

Atom B at t=0.5 and 5 ps

Summary of Simulation for metallic systems

: Gold helical multishell nanowire

Two-stage model of gold helical multishell nanowire

(1) Effectivity of NRL Tight-binding Hamiltonian

(2) Nano-science

(1) Two stage model

- a. Dissociation of atoms on the outermost shell from inner one.
- b. **Slip deformation** of atoms row with generating (111) surface.

(2) Competition between s- and d-electrons.

(3) Magic number, helicity, multi-shell structure.

Summary for Method of the Electronic Structures in Large Systems

$10^4 \sim 10^7$ atoms (10~100nm)

Tight-binding formulation of the total electronic energy

Generalized Wannier states --- Unitarity freedom of eigen states

- (1) Density matrix represented by Wannier states
- (2) **Insulators / semiconductors**

Krylov Subspace Method

Subspace diagonalization method

Shifted COCG method

- (1) Density matrix in Krylov subspace
- (2) **Insulators / semiconductors and metals**

Hybrid scheme by dividing the Hilbert space
Parallelization

Original method and Program-Code

Problems in future

: Preparation for Hamiltonian

Tight binding Hamiltonian

*** Possibly Outer Loop of LDA Hamiltonian**

(1) Empirical TB Hamiltonian :

[Si, Ge, C: sp^3 model] Kwon et al., Phys. Rev. B**49**, 7242 (1994). cf. sp^3s^* model

(2) Non-orthogonal TB Hamiltonian (NRL) :

[Au, Cu, wide range of elements]

M. J. Mehl and D. A. Papaconstantopoulos, Phys. Rev. B**54**, 4519 (1996),
F. Kirchhoff et al., Phys. Rev. B**63**, 195101 (2001).

(3) Orthogonal/Non-orthogonal TB Hamiltonian with repulsive potentials

Xu et al., J. Phys. Cond. Matt. **4**, 6047 (1992).

Bond-order potential with repulsive potentials

Mrovec et al., Phys. Rev. B**69**, 094115 (2004).

(4) Gaussian or atomic orbital base (Gas phase parameters)